



# Computational design of new multifunctional materials

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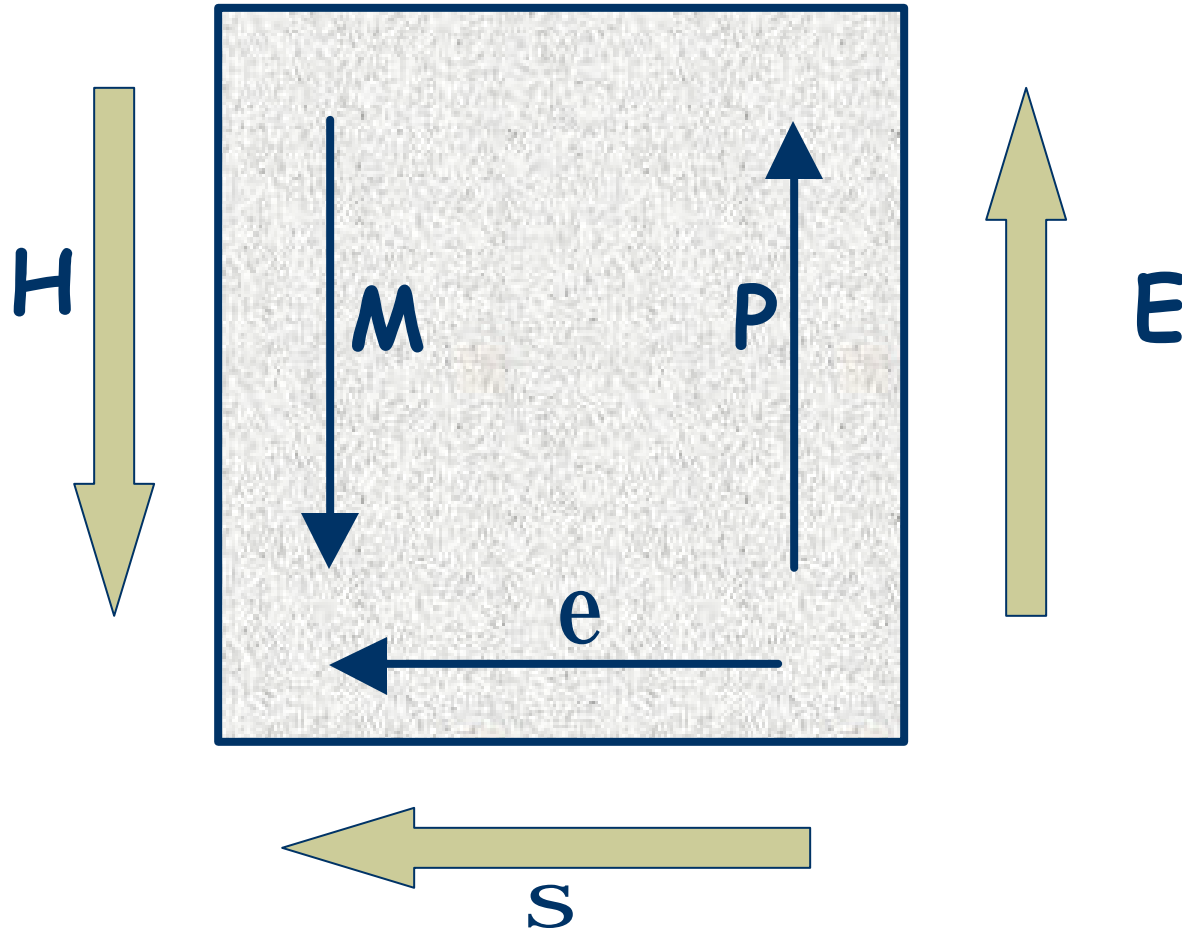
\$\$\$s:

NSF DMR/ITR





# Magnetolectric multiferroics



Review: N.A. Hill, *Ann. Rev. Mat. Res.* **32**, 1-37 (2002).

NSF/ITR workshop, UIUC 2004

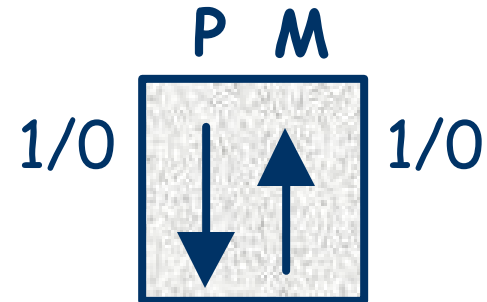


# Possible multiferroic applications?



## Device applications (long range):

- Multiple state memory elements
- Write to  $E$  / read from  $M$
- High  $\mu$ , high  $\epsilon$  materials
- $E$  tunable magnetic properties (and vice versa)



## Recent observations:

- Giant field-tunable non-linear optical response
- Large magnetocapacitance

## Fundamental physics:

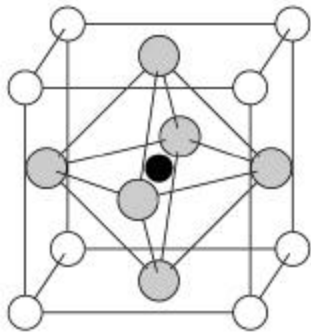
- Nature of coupling between order parameters



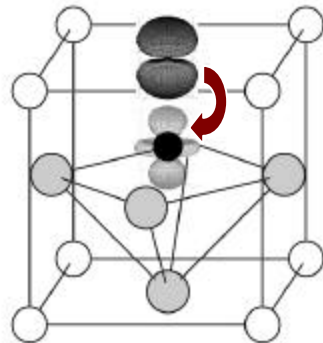
# Hard to achieve (almost none exist)



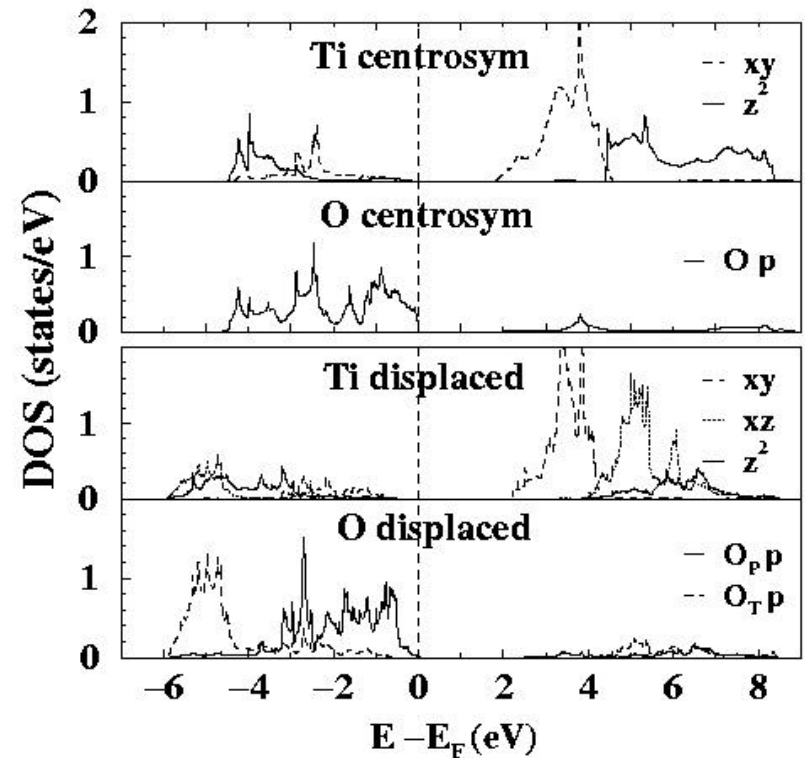
Conventional ferroelectrics have empty cation d orbitals:



paraelectric



ferroelectric





# BUT magnetism requires localized electrons!



In perovskite-structure oxides the source of magnetic, localized electrons is usually the transition metal d electrons, e.g.  $\text{LaMnO}_3$ ,  $\text{SrRuO}_3$ , etc.

**BAD NEWS!**

Ferromagnetism requires d electrons

Ferroelectricity requires "d<sup>0</sup>-ness"

**CHEMICALLY INCOMPATIBLE!**

Why are there so few magnetic ferroelectrics?

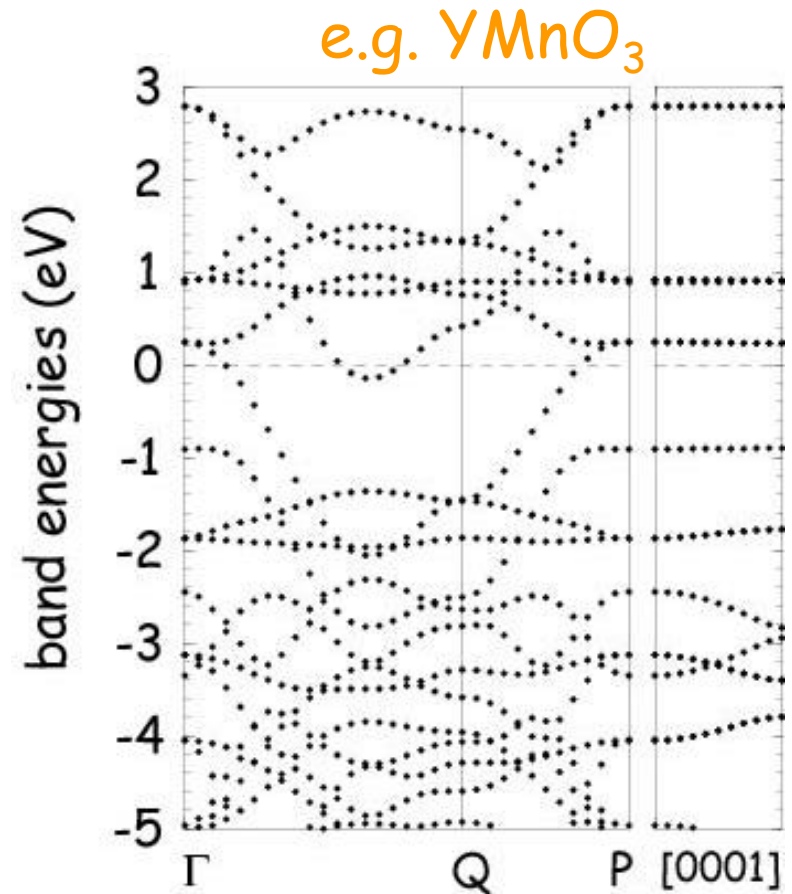
N.A. Hill, J. Phys. Chem. B 104, 6694-6709 (2000)



# ... and hard to calculate



**LSDA?** Usually OK for ferroelectrics, but not for **MAGNETIC** ferroelectrics! Often obtain metallic band structures, prohibiting the calculation of ferroelectric polarization.



Beyond-LDA methods, **LDA+U** or Self-Interaction-Corrected (**SIC**), are needed...



# Self-Interaction



The self-interaction is the interaction of an electron's charge with the Coulomb and exchange-correlation potential generated by the same electron.

Consequences:

**Underestimated:**

- binding energies
- on-site Coulomb energies (Hubbard  $U$ )
- exchange splittings of d and f states

**Overestimated:**

- anion p - cation d hybridizations
- corresponding band widths ( $W$ )

Suppression of  $U$  and overestimation of  $W$  is a problem for materials with partially filled d states where, in real life,  $U \gg W$



# Our choice: pseudo-SIC method - subtract off the exchange and correlation self-interaction within a pseudopotential formalism



A. Filippetti and N.A. Spaldin, *Self-interaction corrected pseudopotential scheme for magnetic and strongly correlated systems*, *Phys. Rev. B* **67**, 125109 (2003).

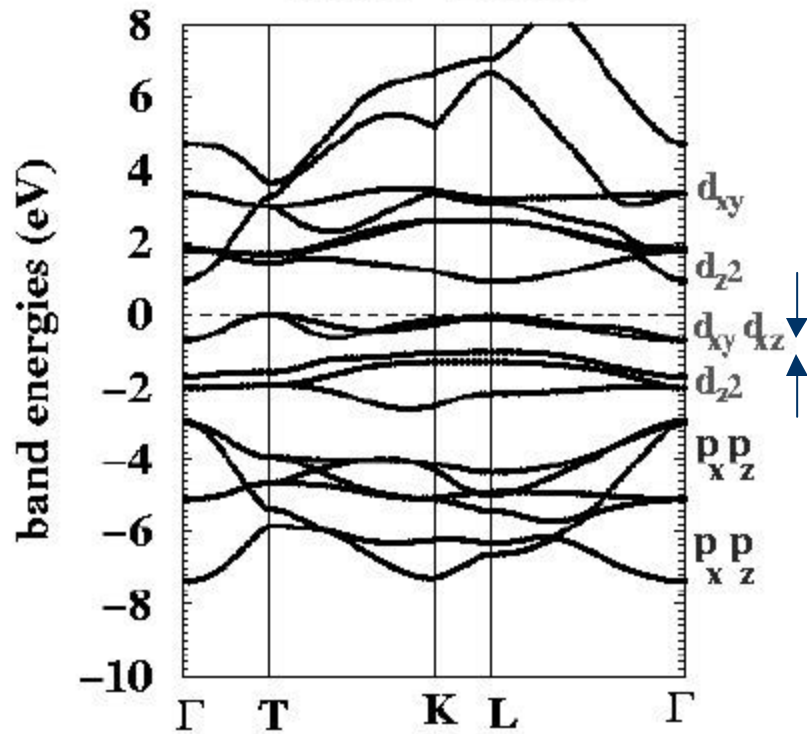


## Builds on:

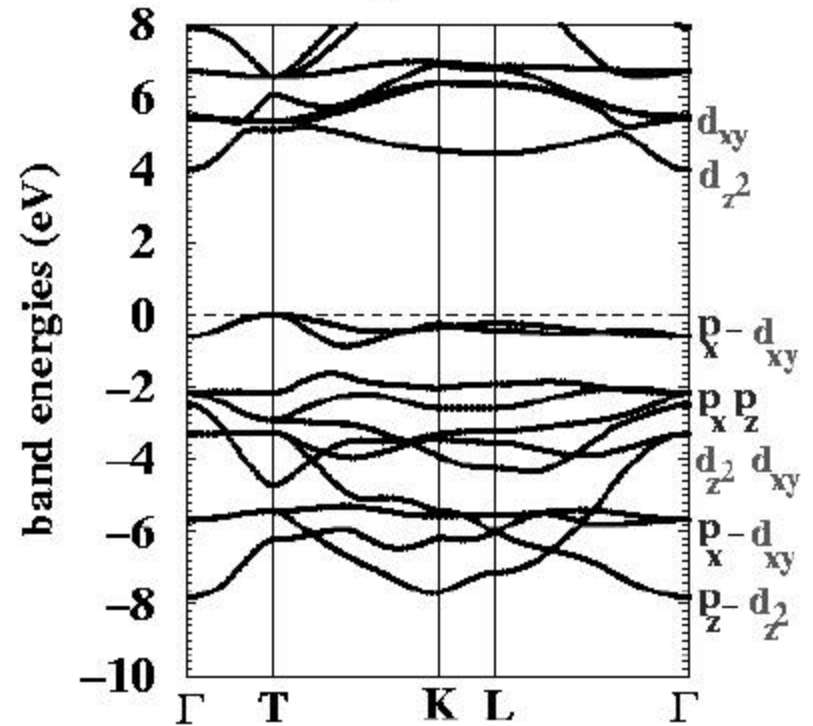
- ◆ Perdew and Zunger, *PRB* **23**, 5048 (1981). Extensive discussion and successful application to atoms and molecules.
- ◆ Svane et al., 1994 - present. Application of fully self-consistent SIC to solids. (impressive but expensive! LMTO implementation).
- ◆ Vogel et al., 1996-98. SIC pseudopotentials used in regular LDA calculation. Successful and cheap.



MnO LSDA



MnO pseudo-SIC



- gap too small and wrong character (Mott-Hubbard)
- incorrect d-p splitting
- magnetic moment too small

- ✓ strong d-p mixing
- ✓ no d-p splitting
- ✓ large gap
- ✓ intermediate charge-transfer/ Mott Hubbard regime
- ✓ correct magnetic moment

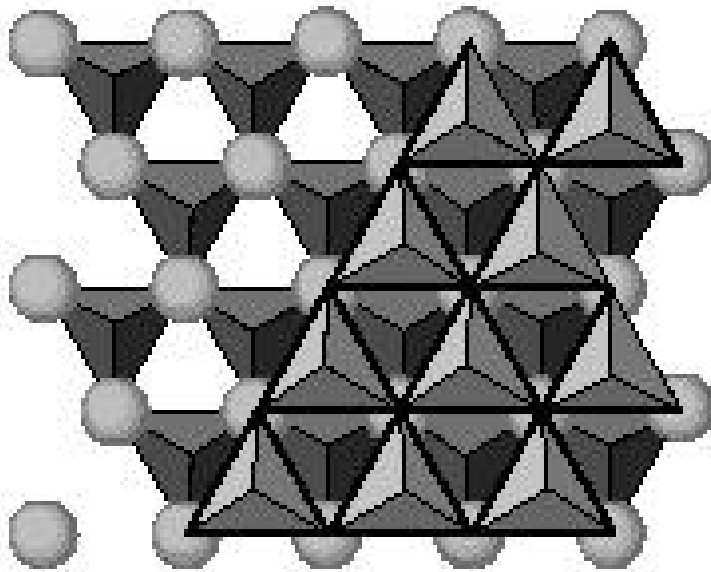
NOTE: pseudo-SIC results are very similar to LDA+U!



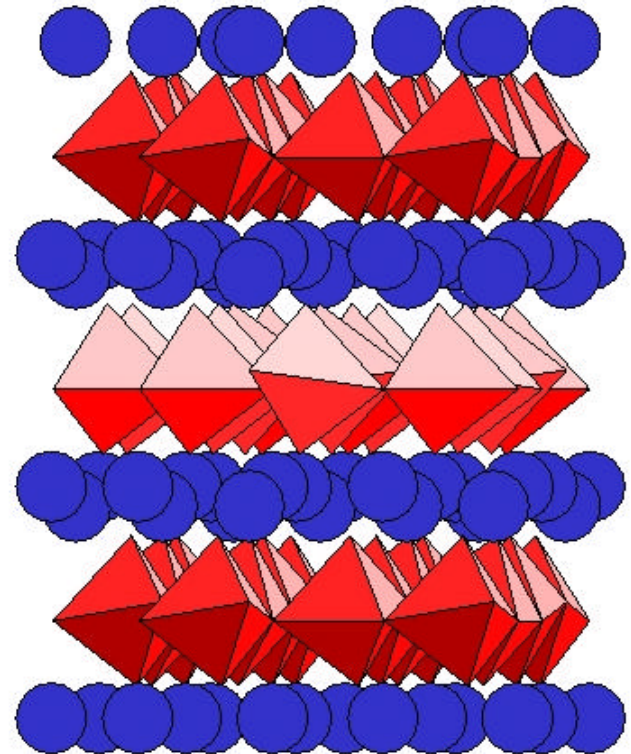
# An anomalous ferroelectric: $\text{YMnO}_3$



View down  $c$ , above  $T_c$



Side view, below  $T_c$



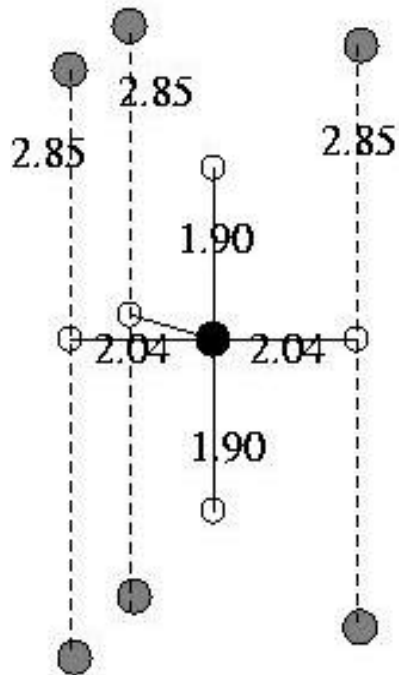
What is the origin of the ferroelectricity?



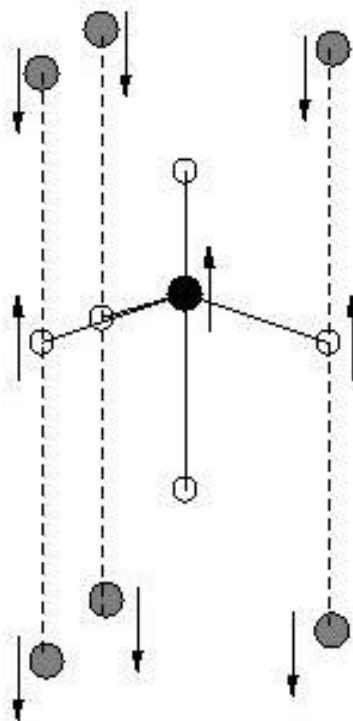
# Explore possible instabilities computationally:



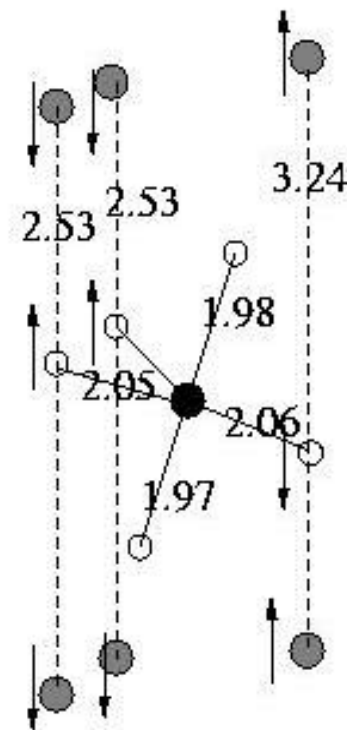
centrosymmetric



unfavorable

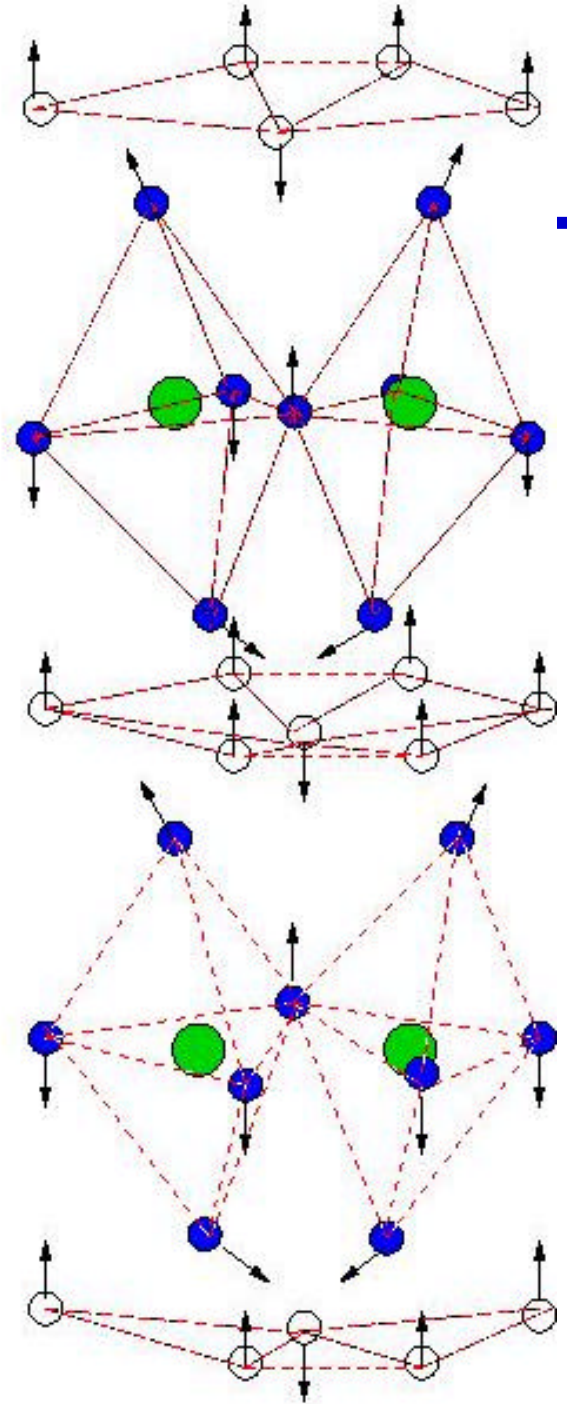
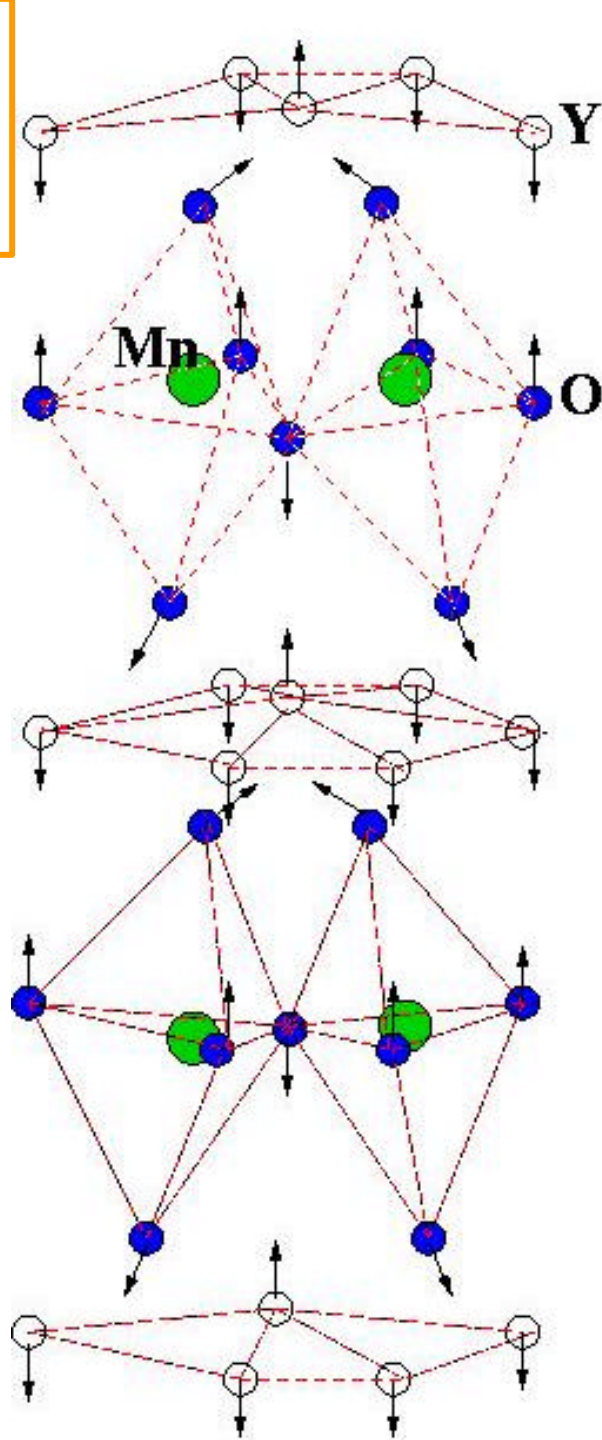


favorable



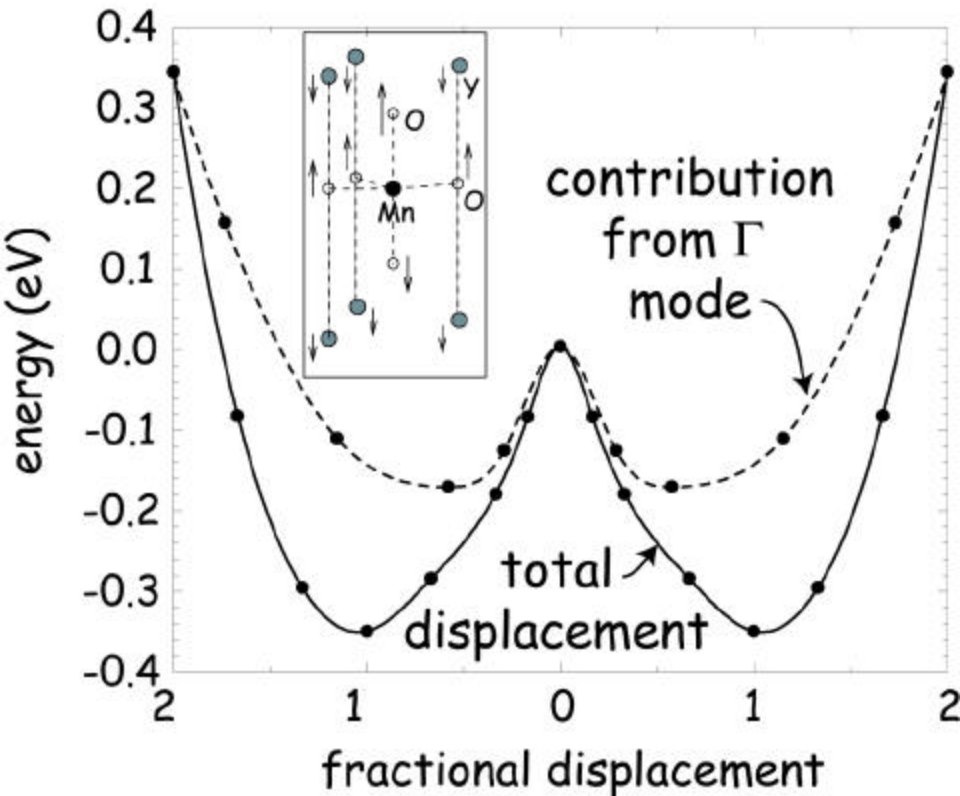
- Y
- Mn
- O

Two stable states:





# Geometrically-driven ferroelectricity



- 1) Buckling of layered MnO<sub>5</sub> polyhedra (P63/mmc to P63cm)
- 2) Relative Y-O motion along c axis (no further symmetry lowering)
- 3) Driven by electrostatic and size effects
- 4) PERMITS COEXISTENCE OF FERROELECTRICITY AND MAGNETISM!
- 5) Two phase transitions since confirmed experimentally

The origin of ferroelectricity in magnetoelectric YMnO<sub>3</sub>, B.B. van Aken, T.T.M. Palstra, A. Filippetti and N.A. Spaldin, Nature Materials 3, 164 (2004).



# Usual indicators of instability do not hold:



## DOSs - no re-hybridization

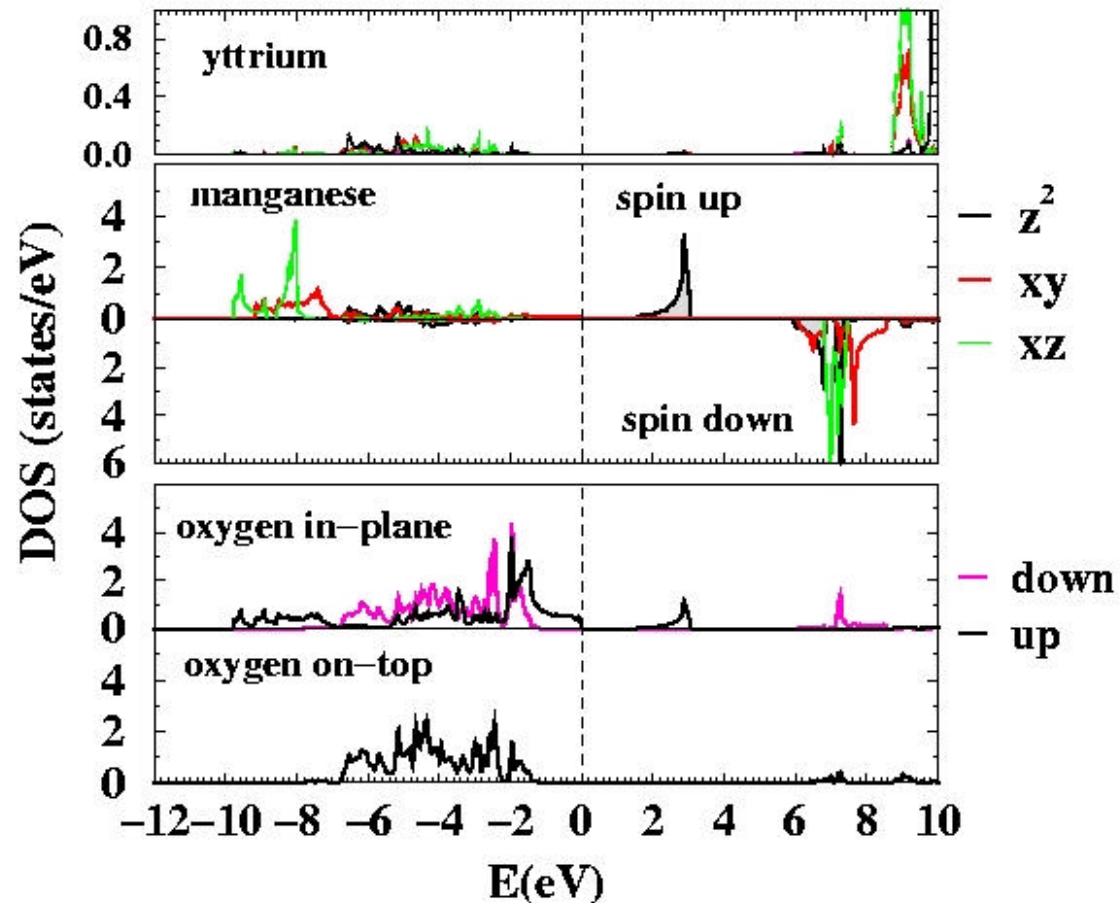
Non-anomalous  
BECs:

Y +3.6

Mn +3.3

O<sub>T</sub> -2.3

O<sub>p</sub> -2.2





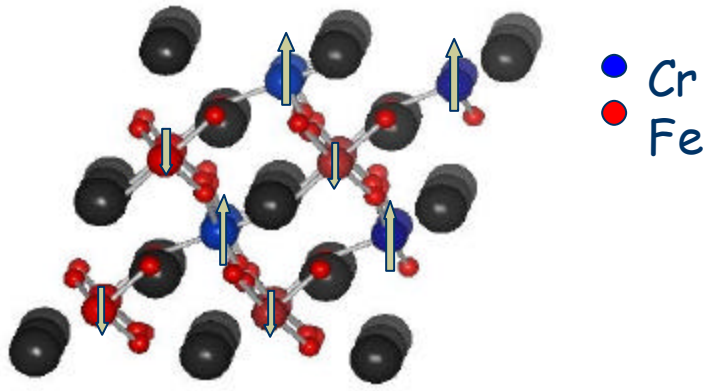
# Ongoing work: Better choices for robust magnetic ferroelectrics:



## Bi-based ferrimagnets (strongly insulating)

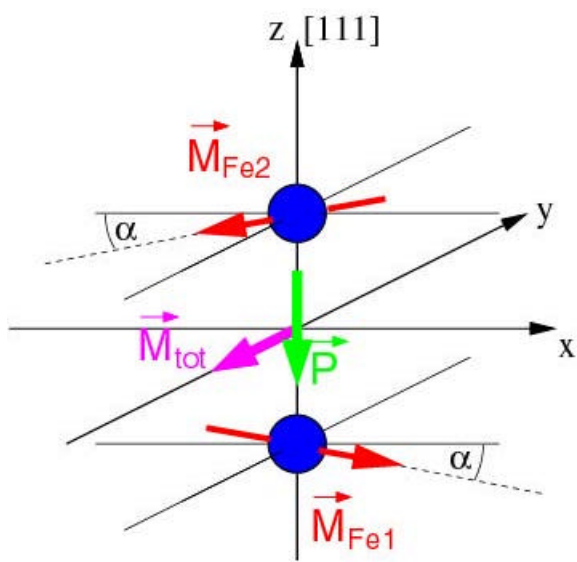
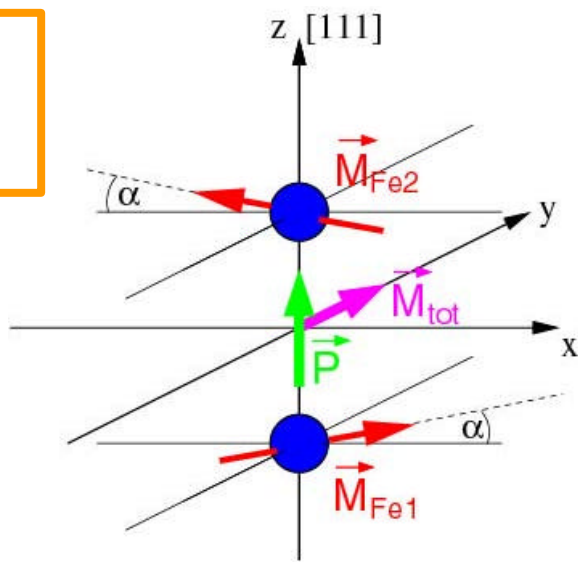
We predict for (111) layered  $\text{Bi}_2\text{FeCrO}_6$

- Magnetic moment =  $2 \mu\text{B}/\text{unitcell}$
- Spontaneous polarization =  $70 \mu\text{C}/\text{cm}^2$



## Weak ferromagnets - coupled M and P!

e.g.  $\text{BiFeO}_3$

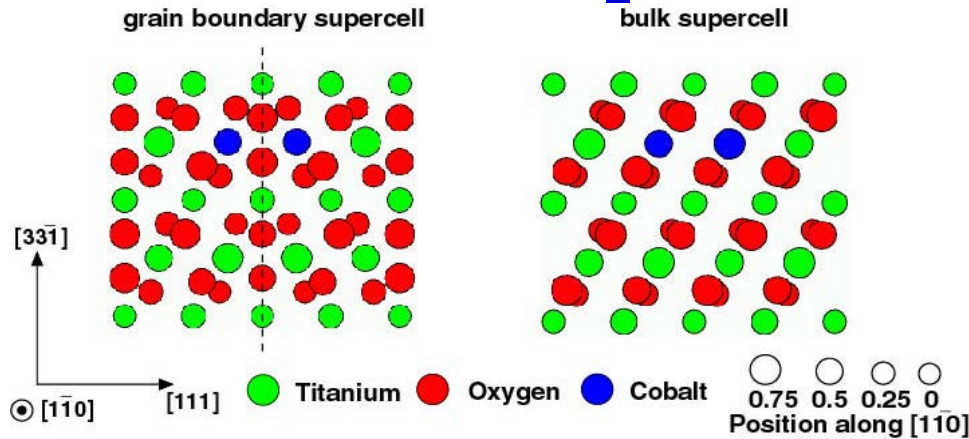




# More ongoing Work

## Grain boundary effects in $(\text{Co,Ti})\text{O}_2$ :

We identified a likely grain boundary in anatase  $\text{TiO}_2$ :  $\Sigma 5$   $(113)[110]$



Now we're calculating the influence on magnetic properties....

## Piezoelectric magnetic semiconductors, e.g. $(\text{Zn,Mn})\text{O}$ :

- ✓  $\text{MnO}$  in wurtzite structure is strongly piezoelectric BUT
- ✗  $(\text{Zn,TM})\text{O}$  needs carriers (preferably holes!) for ferromagnetism



Possible device architecture

$(\text{Zn,Co,Cu})\text{O}$
$\text{ZnO}$
$(\text{Zn,Co,Cu})\text{O}$
$\text{ZnO}$

## Pb-free piezoelectrics, $\text{Bi}(\text{Al,Ga})\text{O}_3$ :

