

ATAT - A software toolkit for modeling coupled configurational and vibrational disorder in alloy systems

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This work was supported by:

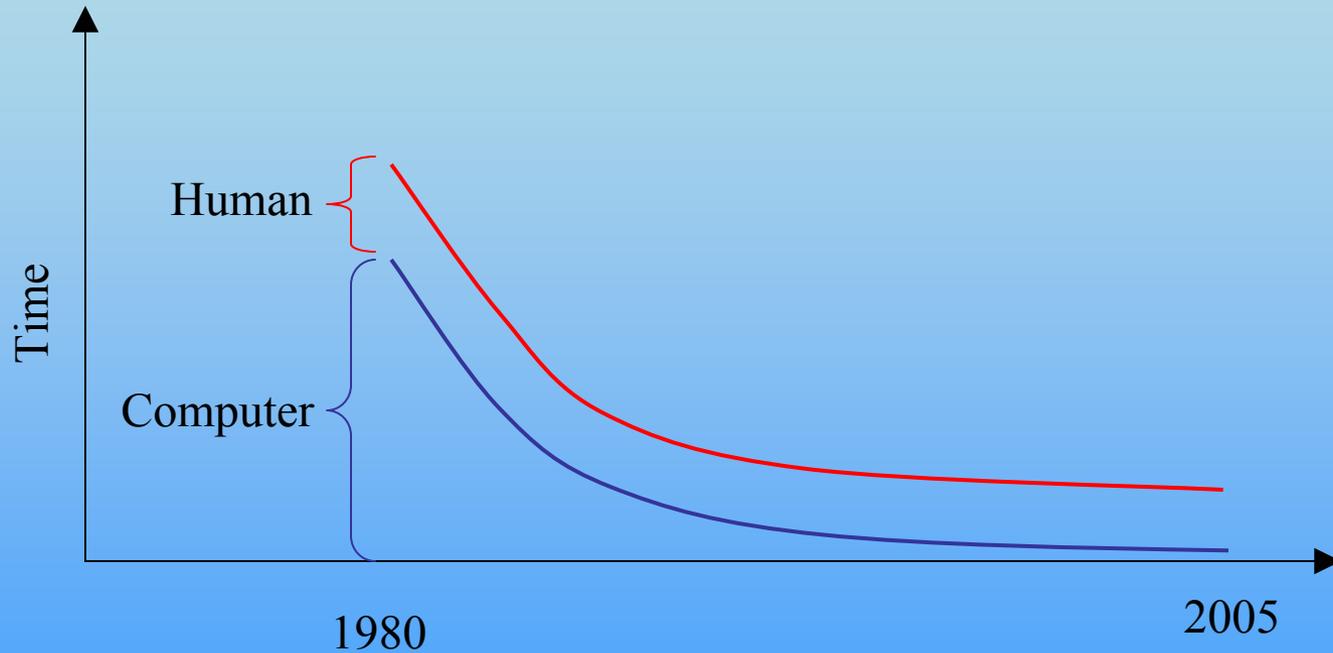
NSF under program DMR-0080766 and DMR-0076097.

DOE under contract no. DE-F502-96ER 45571.

AFOSR-MEANS under grant no. F49620-01-1-0529

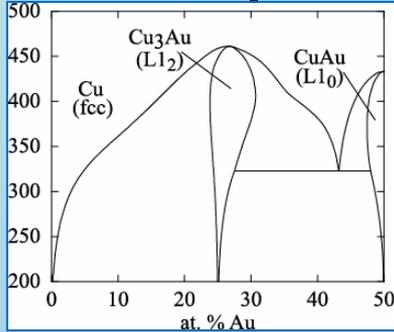
A matter of time...

Time needed to complete a given first-principles calculation

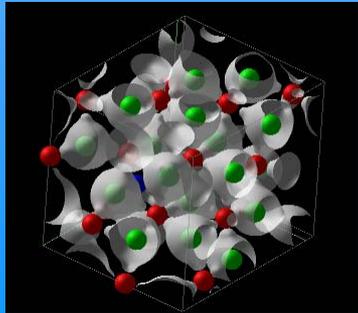
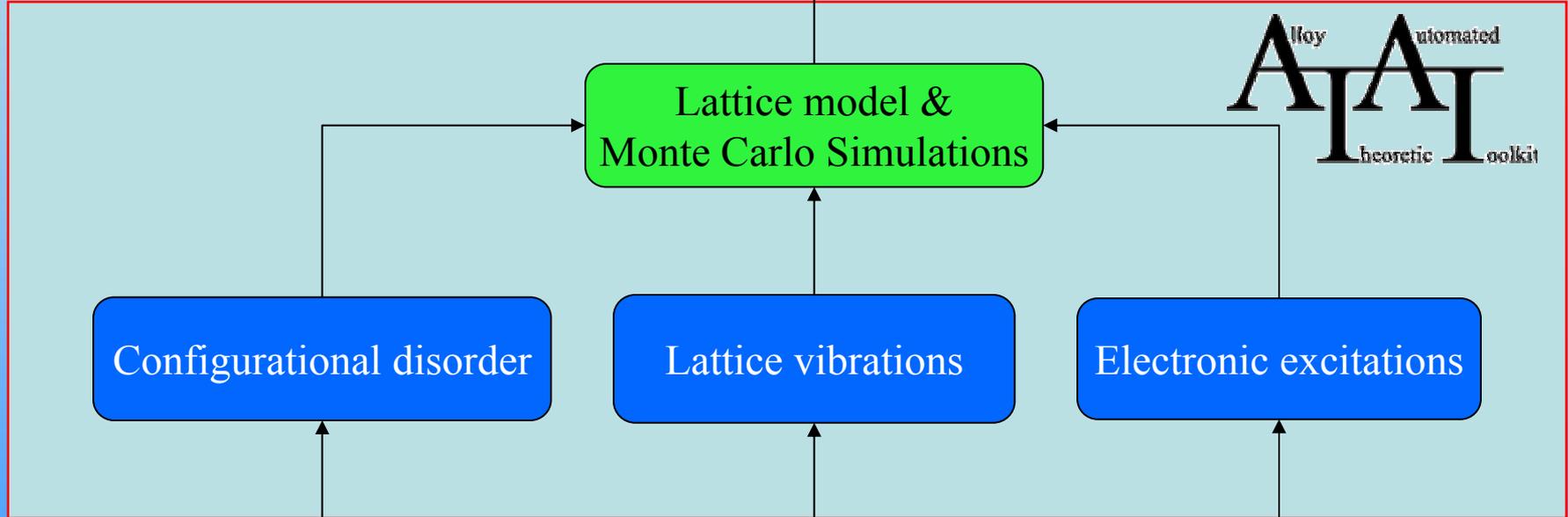


The procedure needs to be automated

First-principles Thermodynamic Calculations



Thermodynamic data



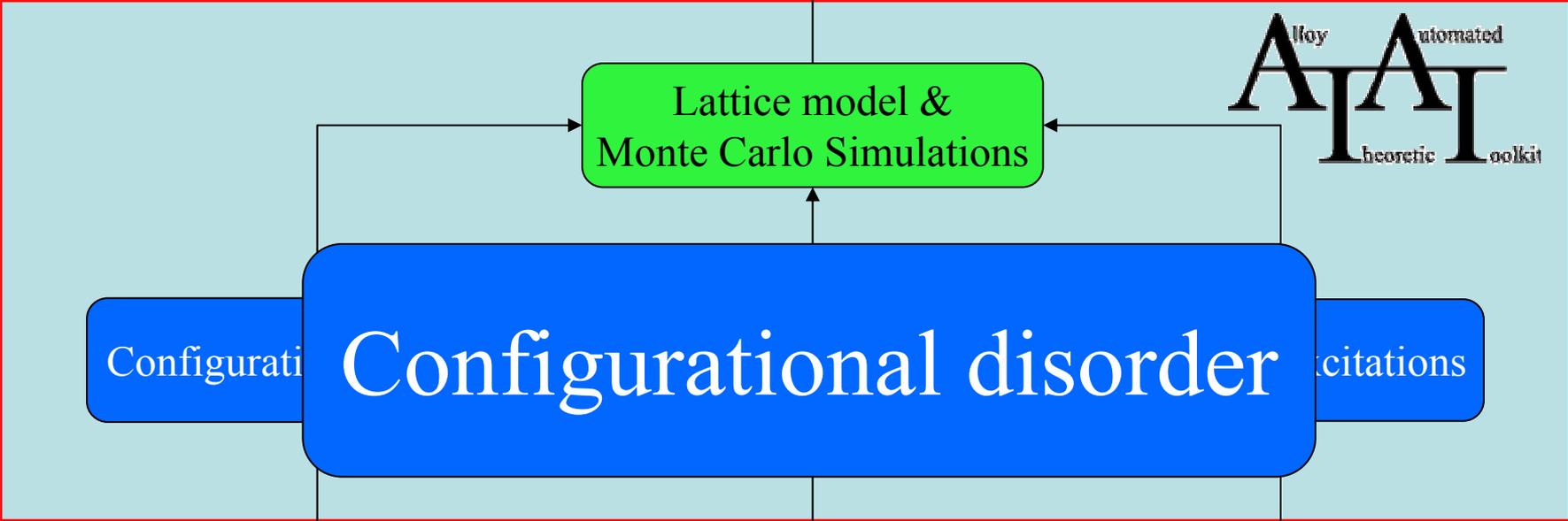
Thermodynamic data

Lattice model &
Monte Carlo Simulations

Alloy Automated
Theoretic Toolkit

Configurati **Configurational disorder** citations

Quantum Mechanical Calculations



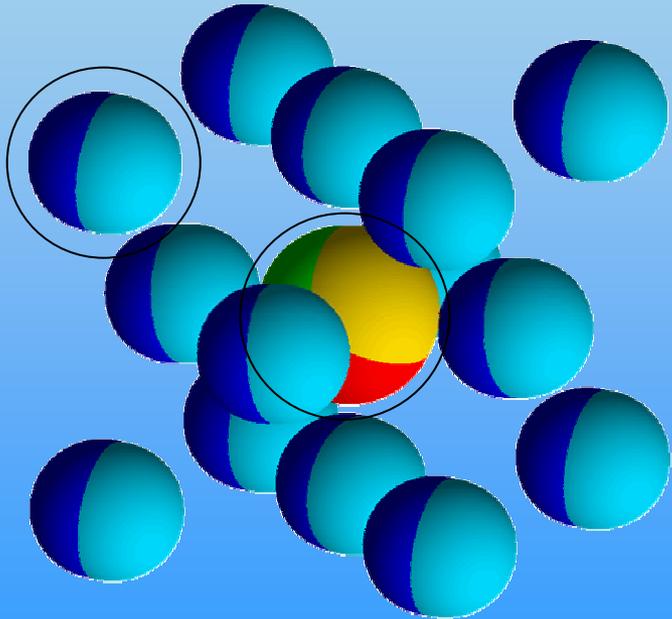
Coupled Sublattices

Multicomponent Cluster Expansion

Same basic form: $E(\sigma_1, \dots, \sigma_n) = \sum_{\alpha} J_{\alpha} \sigma_{\alpha}$

$\sigma_{\alpha} = \prod_i \Gamma(n_i, \alpha_i, \sigma_i)$

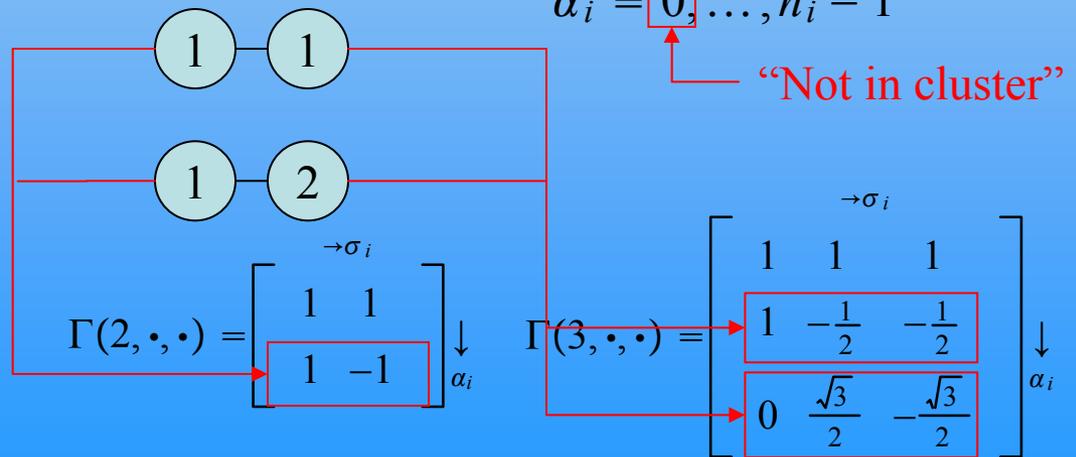
components



Occupation variables: $\sigma_i = 0, \dots, n_i - 1$

“Decorated” clusters: $\alpha = (\alpha_1, \dots, \alpha_n)$

$\alpha_i = 0, \dots, n_i - 1$

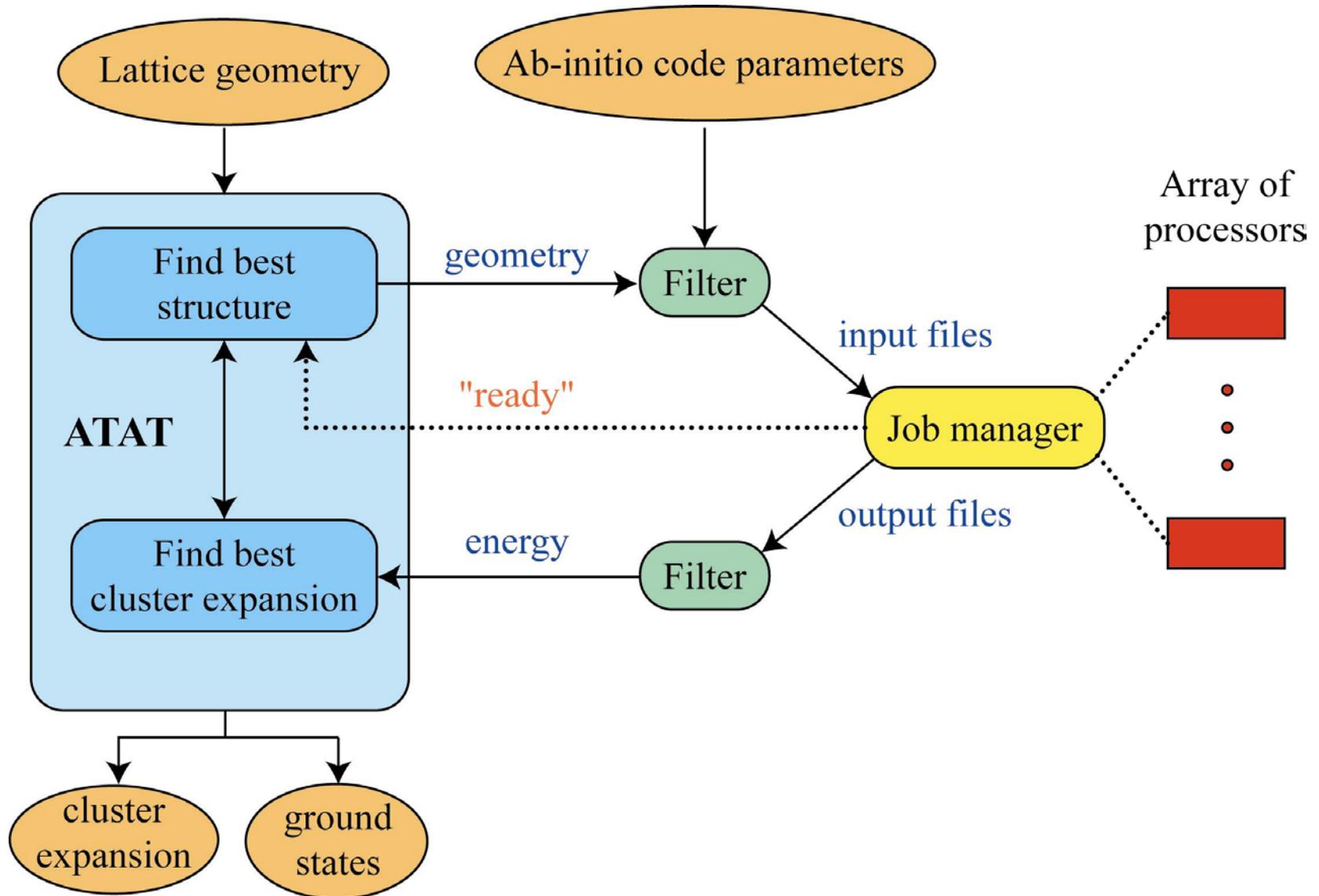


Example: binary fcc sublattice with ternary octahedral sites sublattice

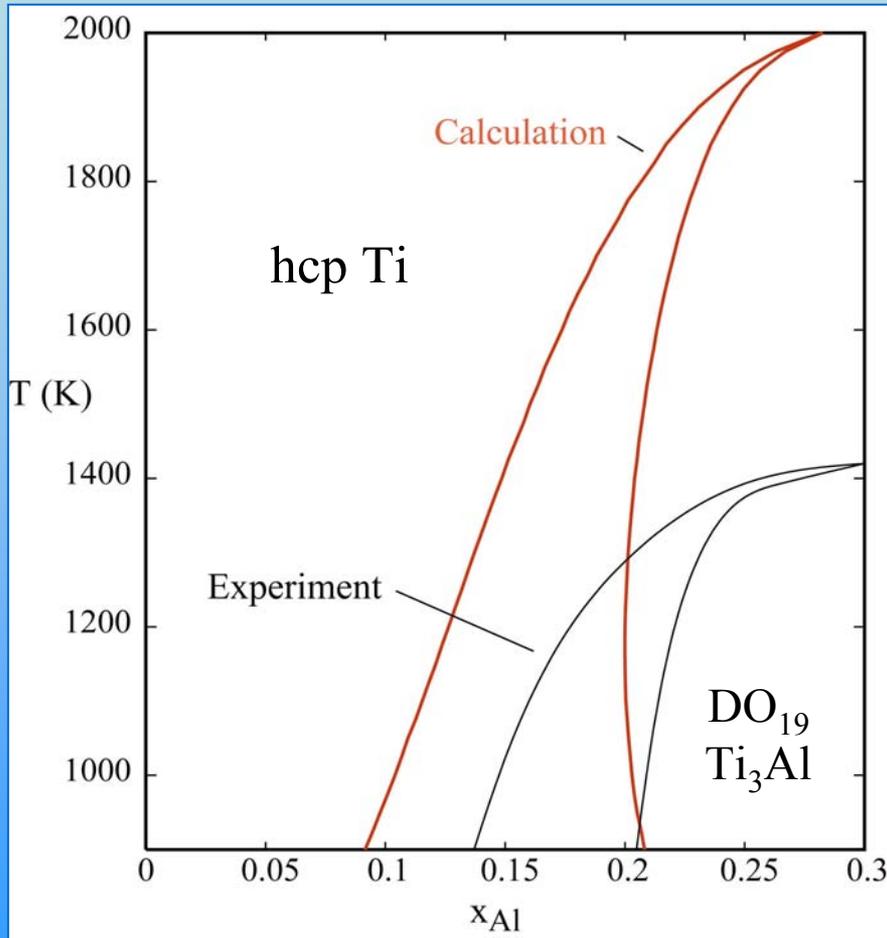
Sanchez, Ducastelle and Gratias (1984)

Tepesch, Garbulski and Ceder (1995)

Automated Cluster Expansion Construction



Temperature scale problem



Likely source of the discrepancy:
Vibrational entropy.

Fultz, Nagel, Antony, *et al.* (1993-1999)
Ceder, Garbulsky, van de Walle (1994-2002)
de Fontaine, Althoff, Morgan (1997-2000)
Zunger, Ozolins, Wolverton (1998-2001)

Van de Walle, Asta and Ceder (2002),
Murray (1987) (exp.)

Many other examples...

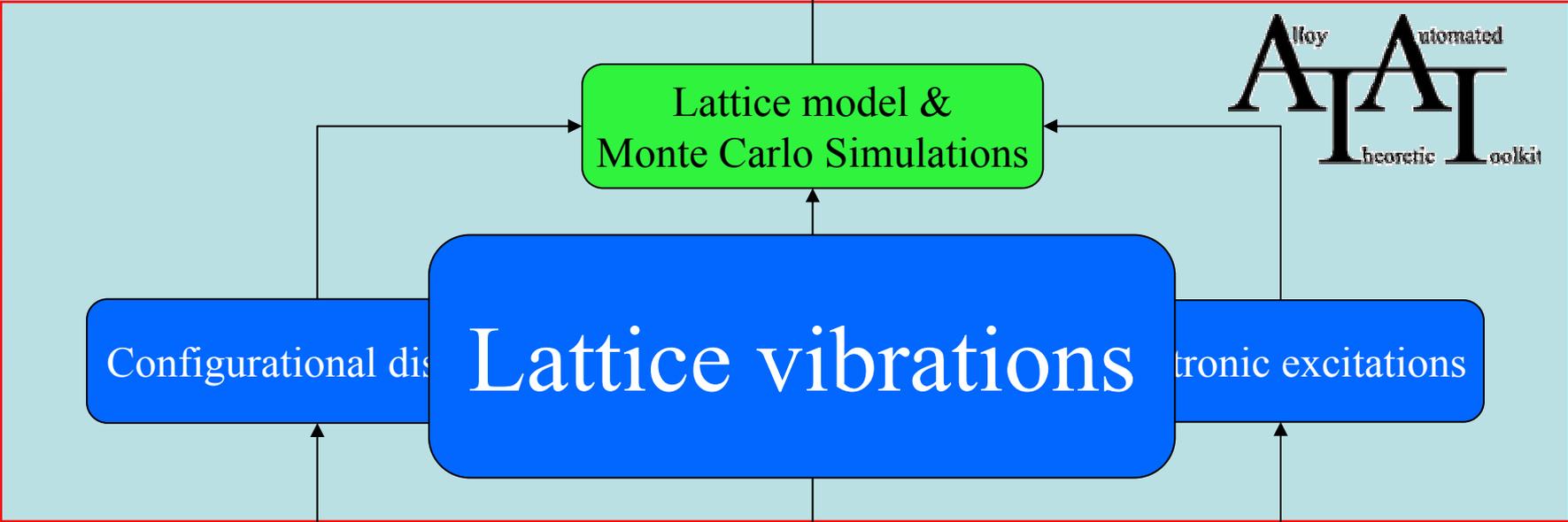
Thermodynamic data

Lattice model &
Monte Carlo Simulations

Alloy Automated
Theoretic Toolkit

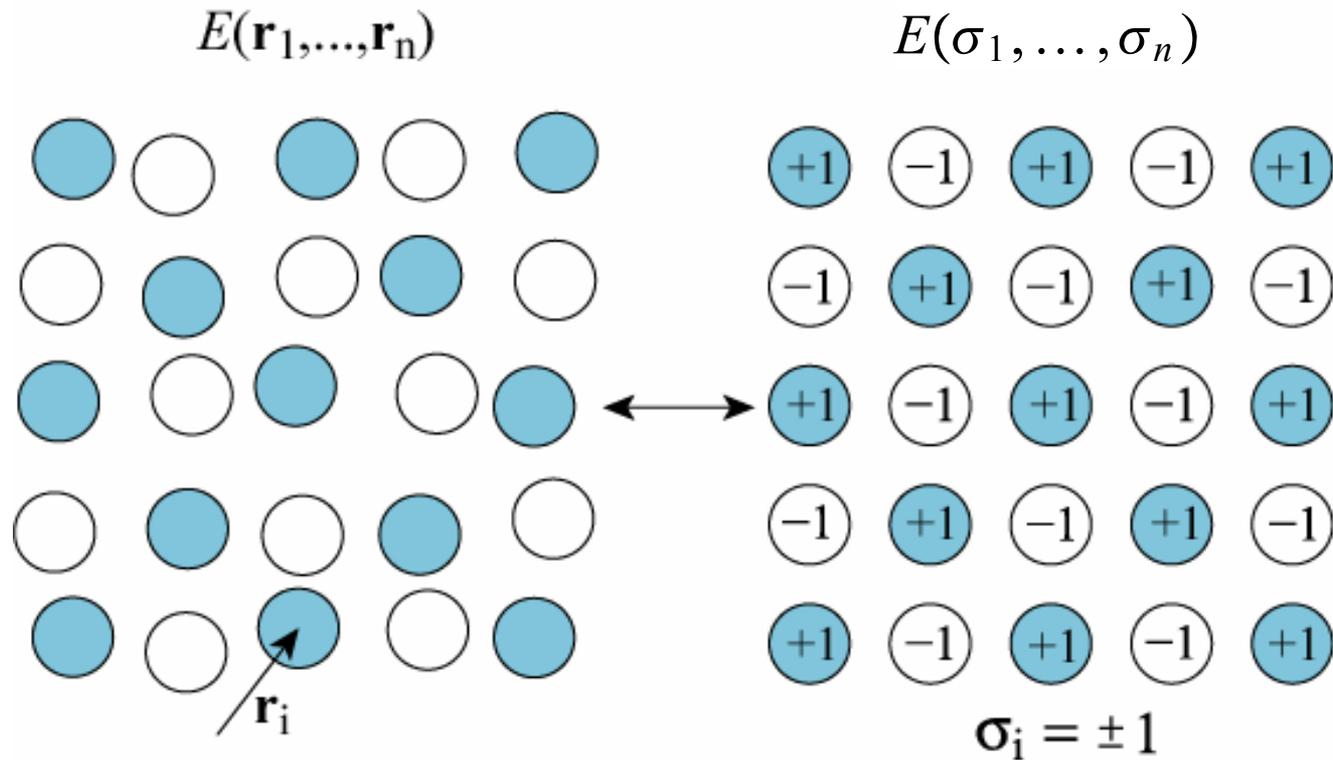
Configurational dis Lattice vibrations Electronic excitations

Quantum Mechanical Calculations



The Cluster Expansion Formalism with Coarse-Graining

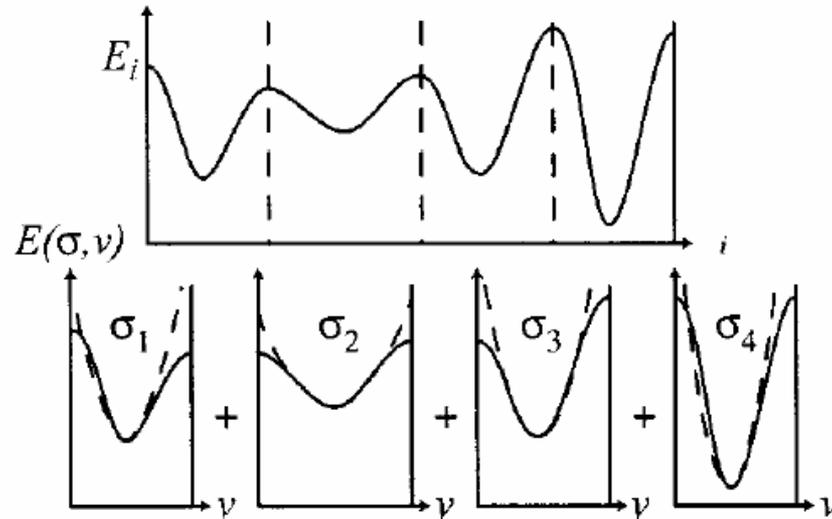
Alloy system \longleftrightarrow Lattice model



$$F(\sigma_1, \dots, \sigma_n) = \sum_{\alpha} J_{\alpha} \boxed{(T)} \sigma_{\alpha}$$

Coarse-Graining of the Free Energy

Graphically:



Formally: (Ceder (1993), Garbulski and Ceder (1994-1996))

$$\begin{aligned} F &= -\beta^{-1} \ln\left(\sum_i e^{-\beta E_i}\right) = -\beta^{-1} \ln\left(\sum_{\sigma} \sum_{i \in \sigma} e^{-\beta E_i}\right) \\ &= -\beta^{-1} \ln\left(\sum_{\sigma} e^{-\beta F(\sigma)}\right) \end{aligned}$$

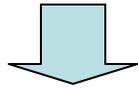
where

$$F(\sigma) = -\beta^{-1} \ln\left(\sum_{i \in \sigma} e^{-\beta E_i}\right)$$

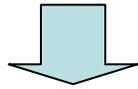
$$\beta = (k_B T)^{-1}$$

First-principles lattice dynamics

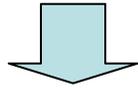
First-principles data



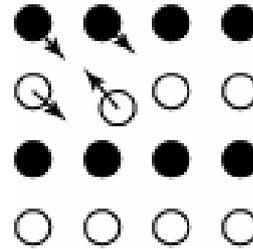
Least-squares fit to Spring model



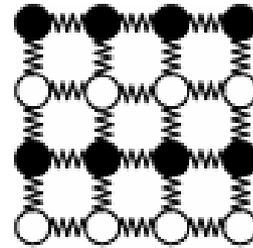
Phonon density of states



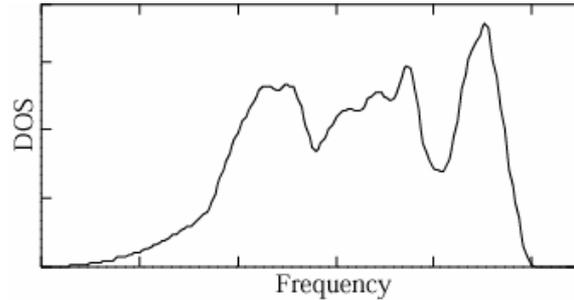
Thermodynamic Properties



Computationally intensive!

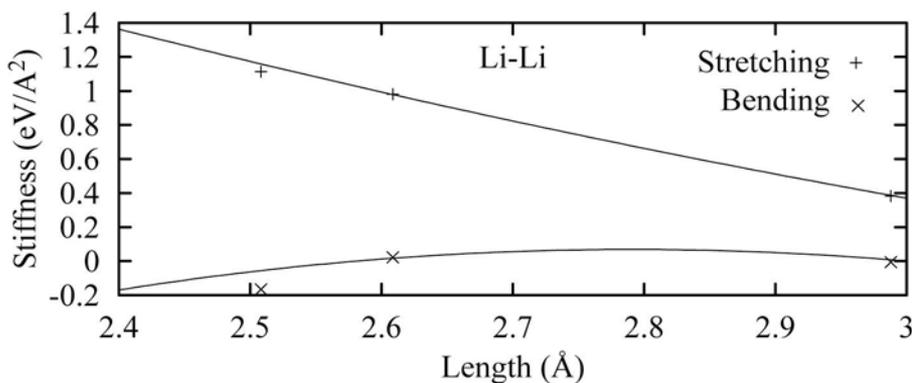
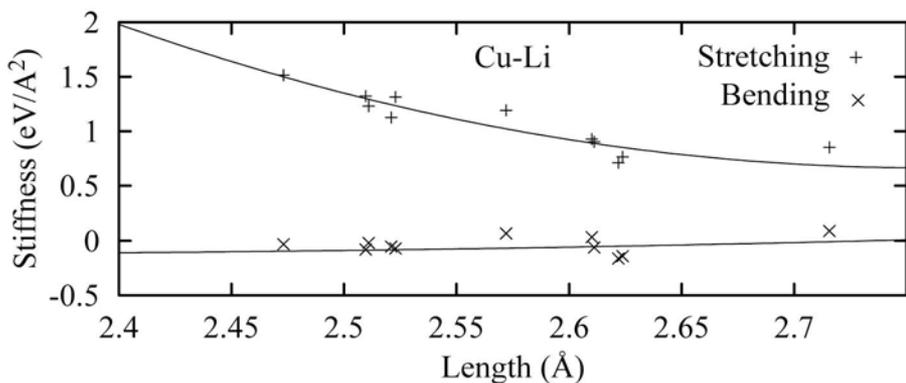
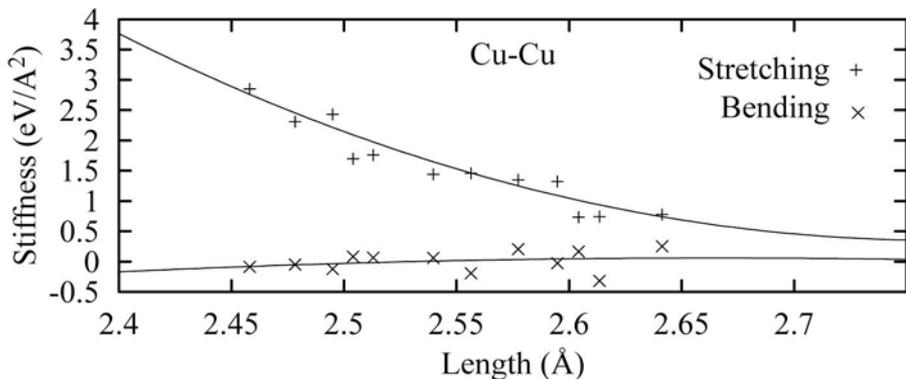


Must be done for many configurations!



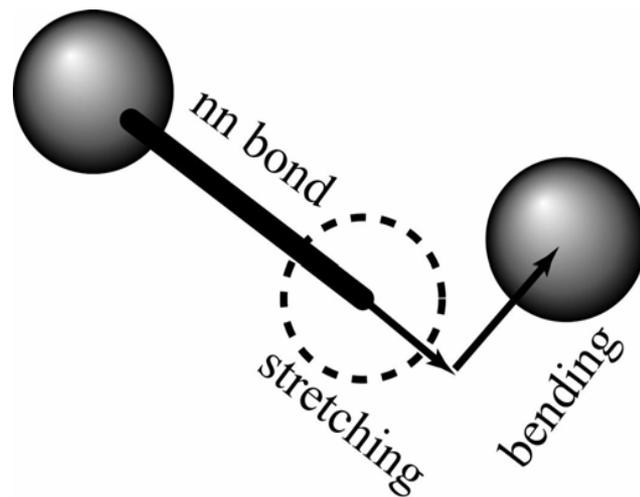
Direct force constant method
(Wei and Chou (1992), Garbuski and Ceder (1994),
among many others)

Transferable Force Constants



Chemical bond type and bond length:

Good predictor of nearest-neighbor force constants (stretching and bending terms)

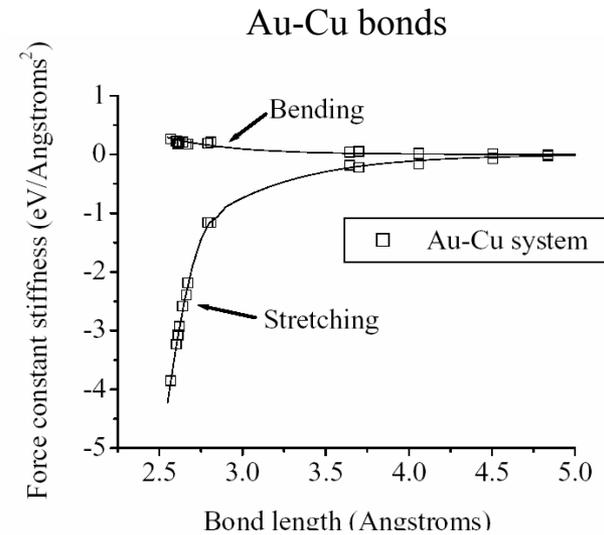
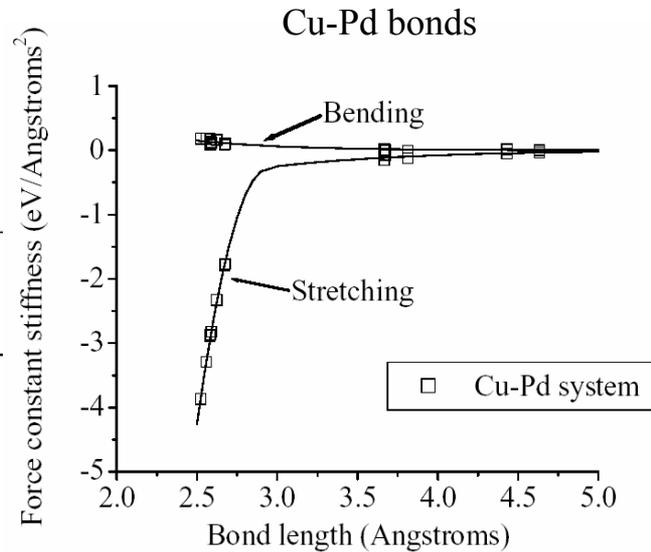
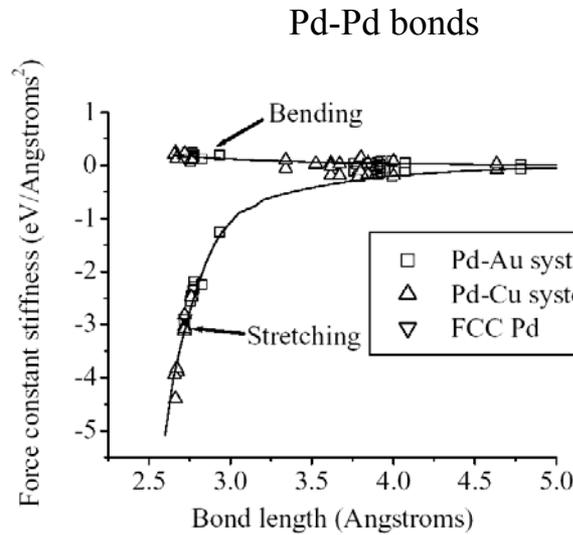
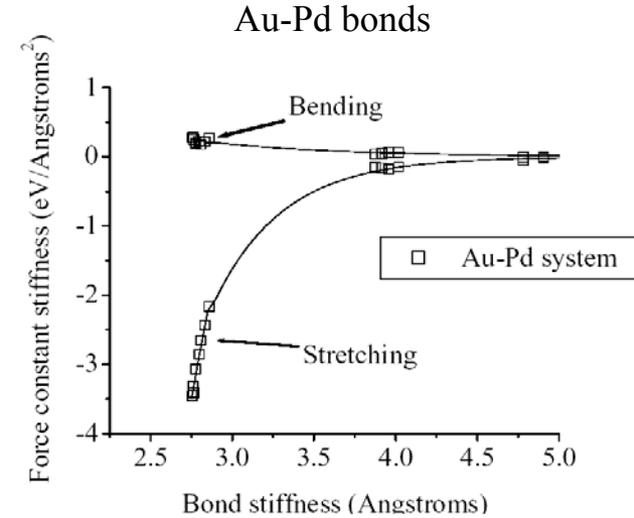
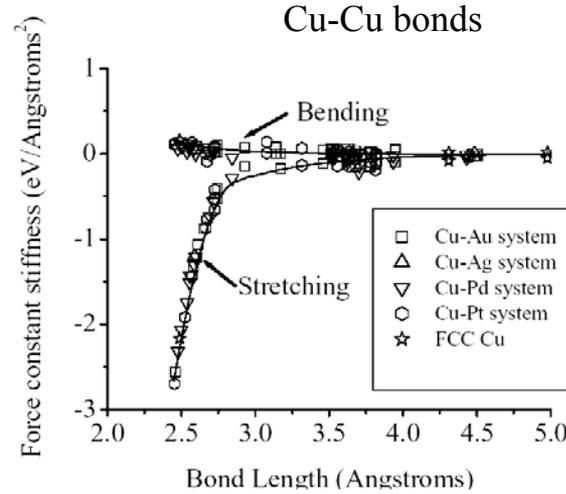
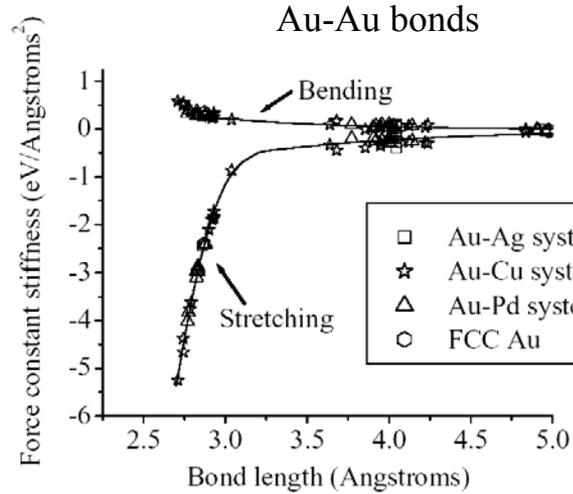


Relationship holds across different structures on the same lattice (here fcc is shown).

van de Walle and Ceder (2000,2002)

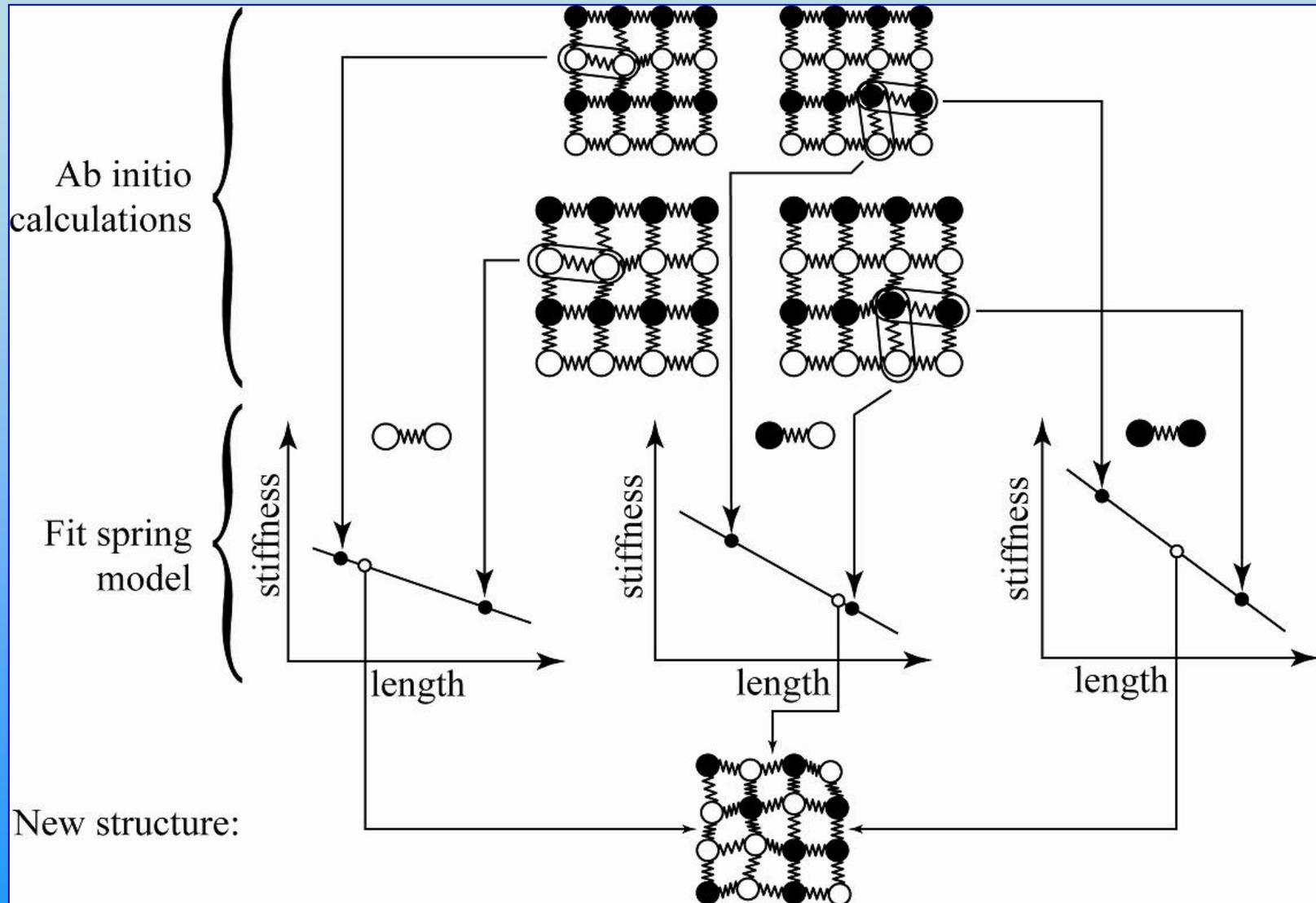
Further tests...

Wu, Ceder, van de Walle (2002)



Accuracy $\sim 0.03 k_B$

Length-Dependent Transferable Force Constants (LDTFC)



Quasi-harmonic model

$$F(T, V) = E(V) + F_H(T, V)$$

Energy of a relaxed motionless lattice
with externally imposed volume V

Vibrational free energy of a harmonic solid
at temp. T with externally imposed volume V

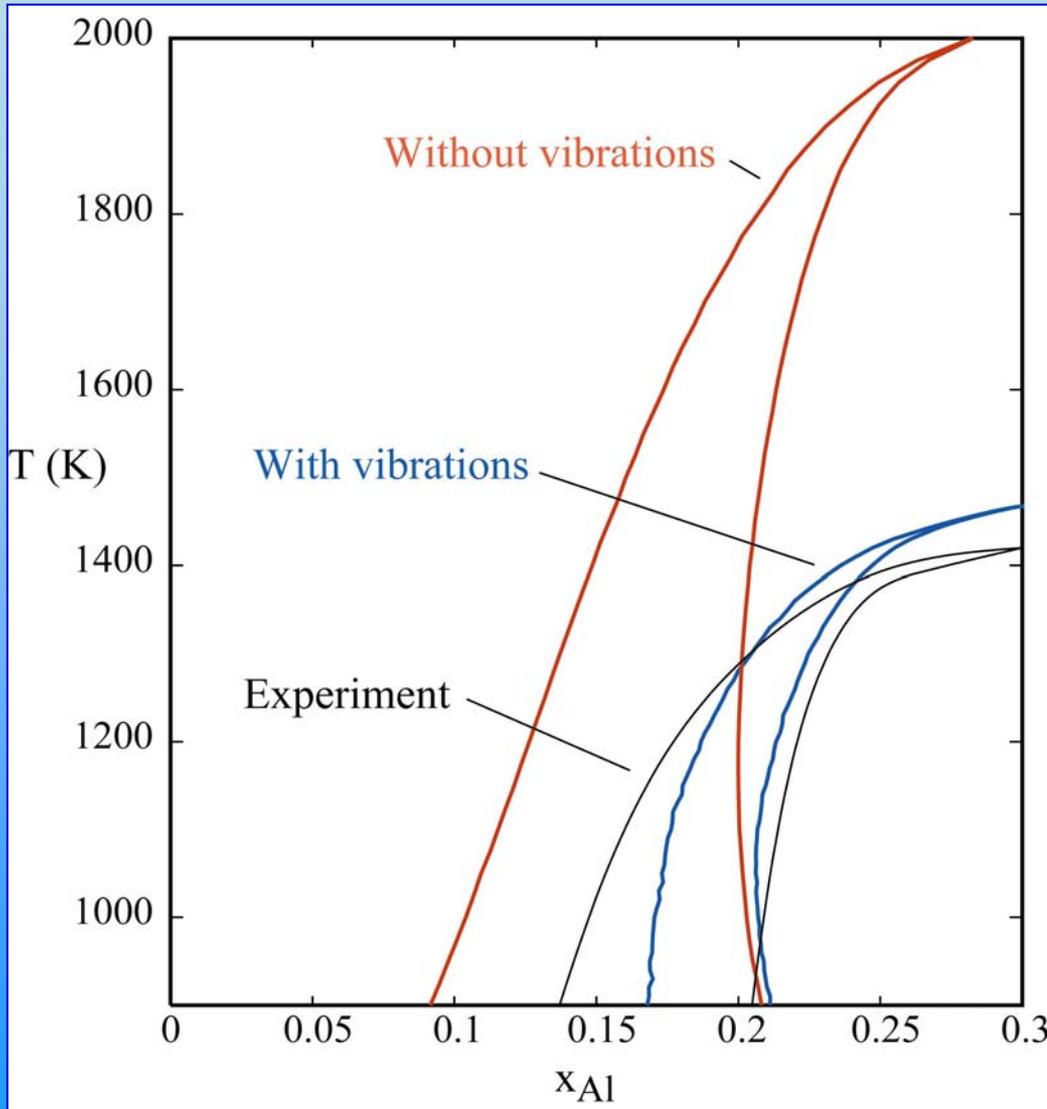
Thermal expansion: $V^*(T) = \arg \min_V F(T, V)$

“True” free energy: $F(T) = F(T, V^*(T))$



Ideal for use with length-dependent transferable force constants

Calculated Ti-Al Phase Diagram



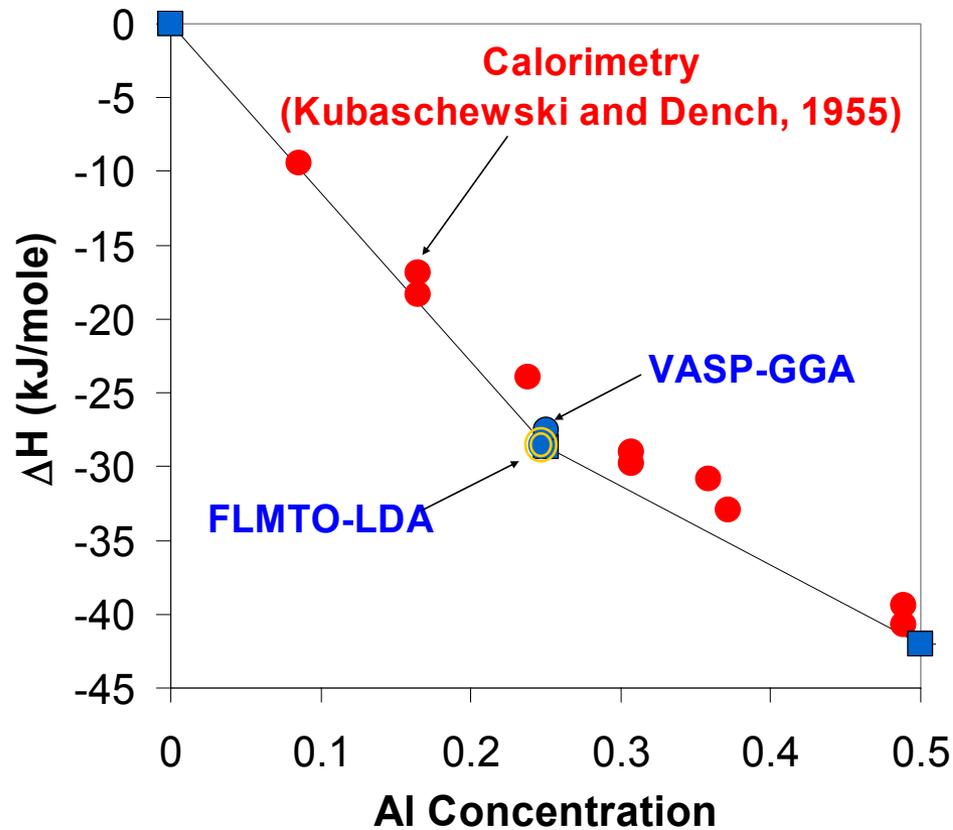
Assessed Phase Diagram:
I. Ohnuma *et al.*, *Acta Mater.* **48**, 3113 (2000)

1st-Principles Calculations:
van de Walle and Asta

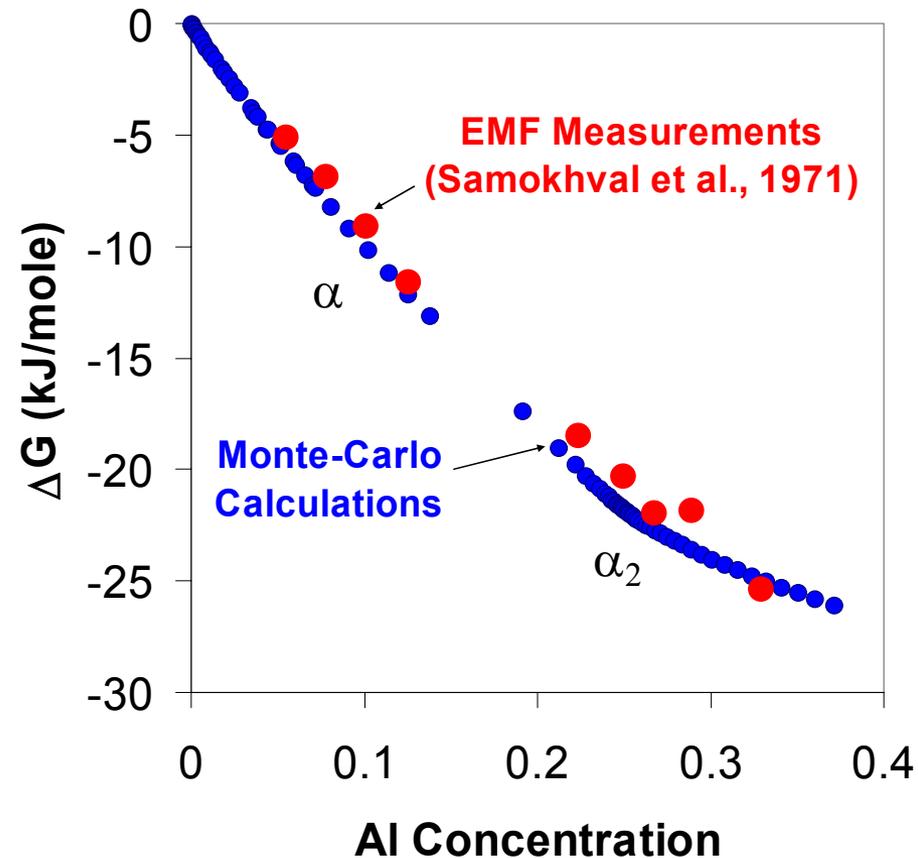
Ti-Al Thermodynamic Properties

1st-Principles Calculations vs. Measurements

Heats of Formation

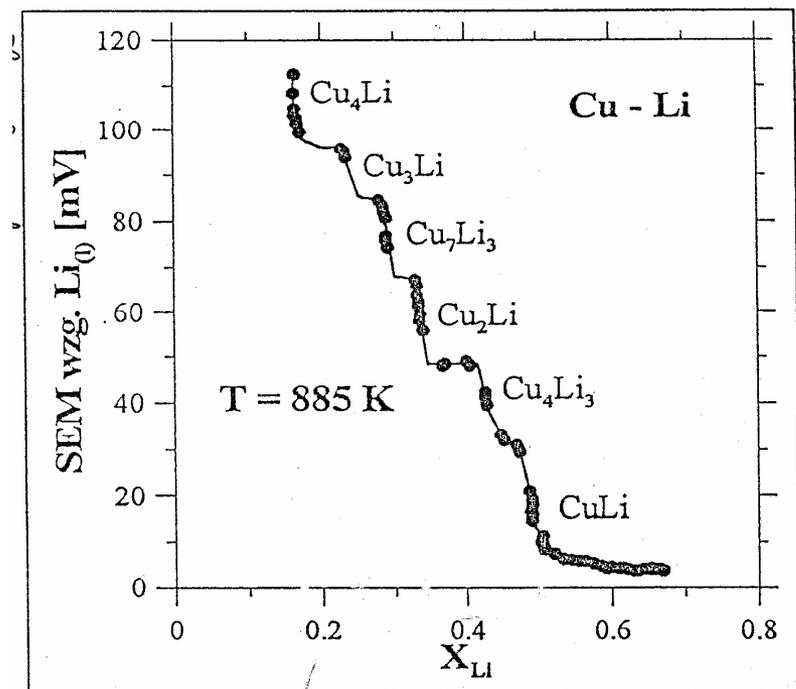
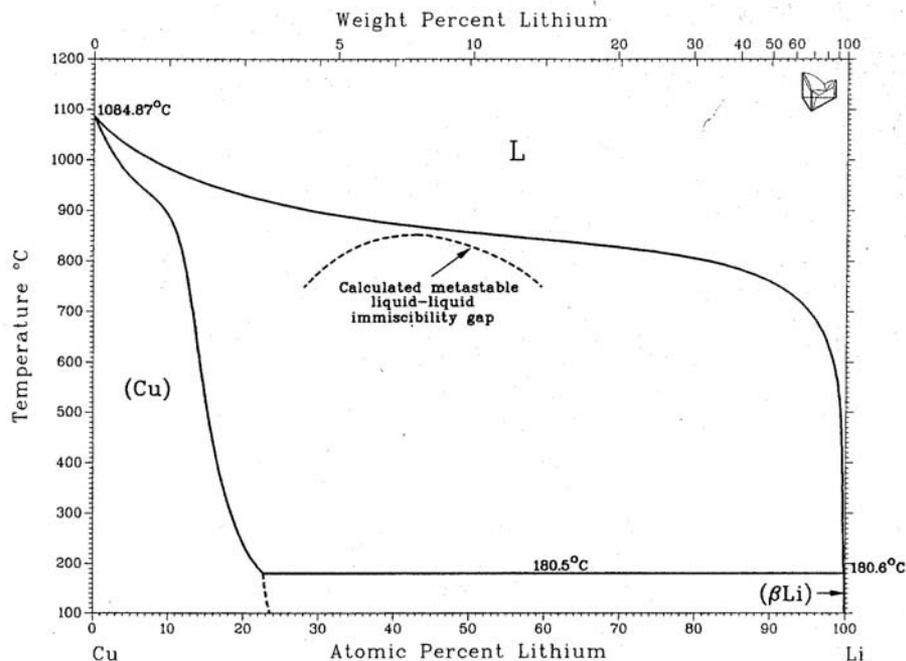


Gibbs Free Energies (T=960 K)



Ordering in the Cu-Li system?

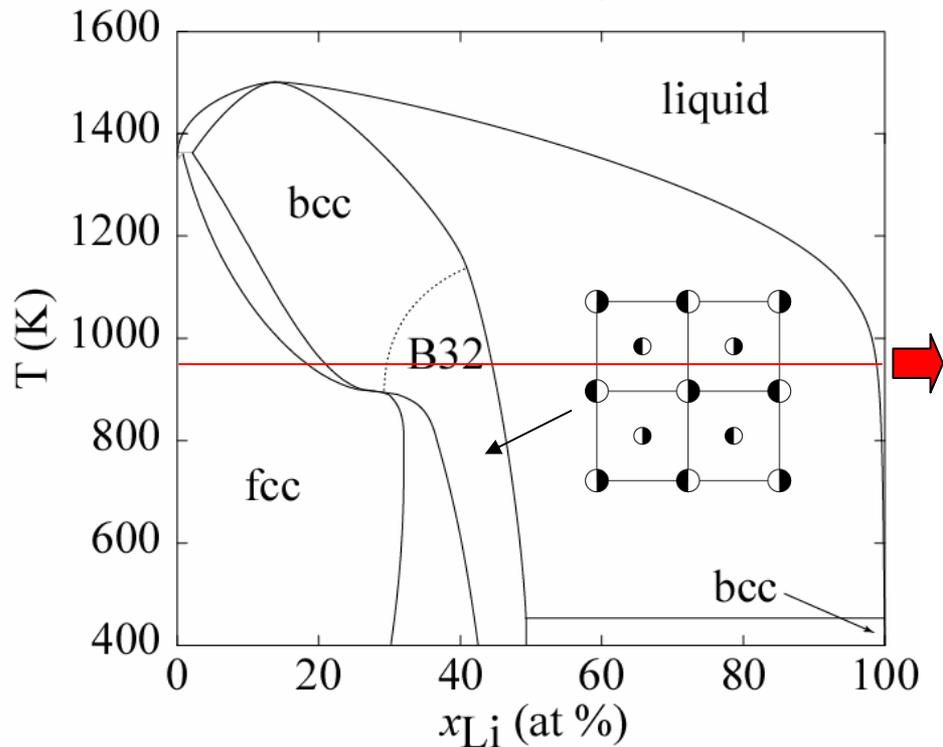
Widely used assessments
do not include ordered phases
(Pelton (1986), Saunders (1998)).



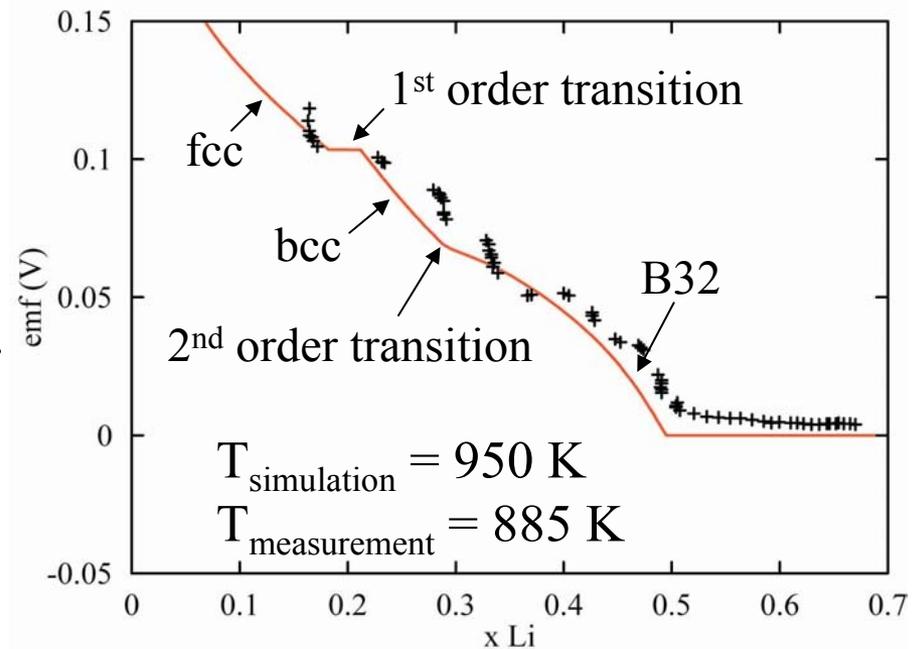
Evidence from EMF measurements
Ordered Cu_4Li phase also suggested by
Borgstedt & Gumiński (1996),
Krauss, Mendelson, Gruen, *et al.* (1986),
Old & Trawena (1981).

Calculated Thermodynamic data for Cu-Li system

Phase diagram



Electromotive Force (EMF)



2 adjustable parameters:

$T_{\text{simulation}}$ + chemical potential of Li (liq)

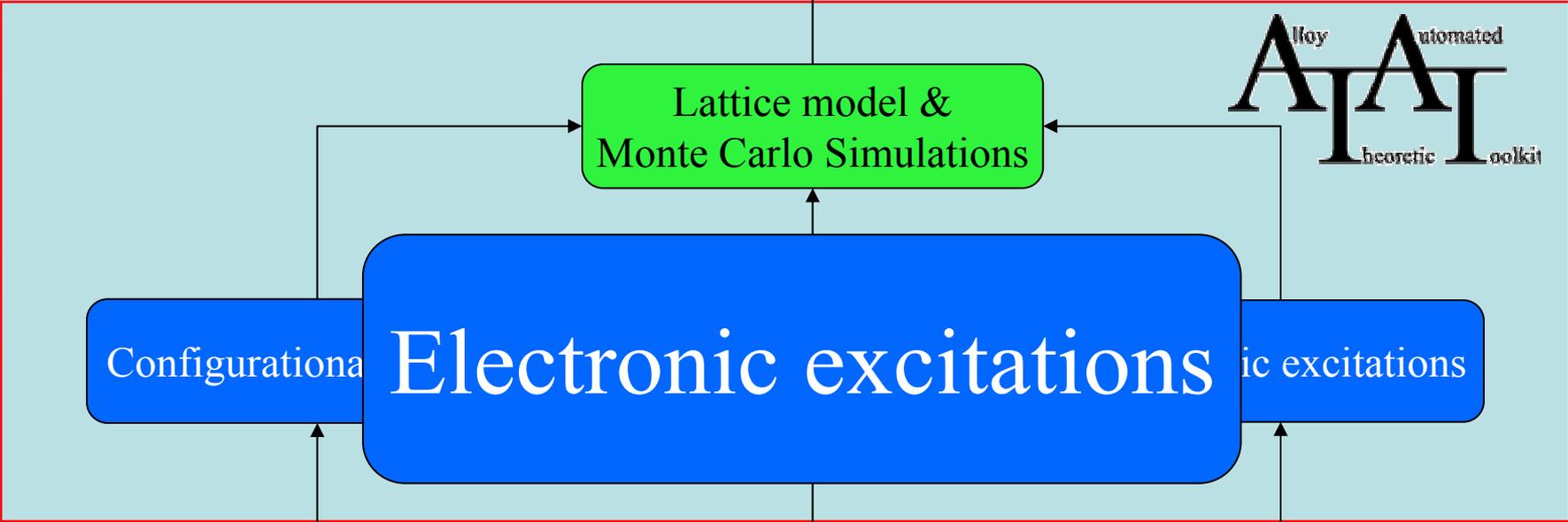
Thermodynamic data

Lattice model &
Monte Carlo Simulations

Alloy Automated
Theoretic Toolkit

Configurational Electronic excitations Electronic excitations

Quantum Mechanical Calculations



Electronic Excitations

Finite-temperature DFT

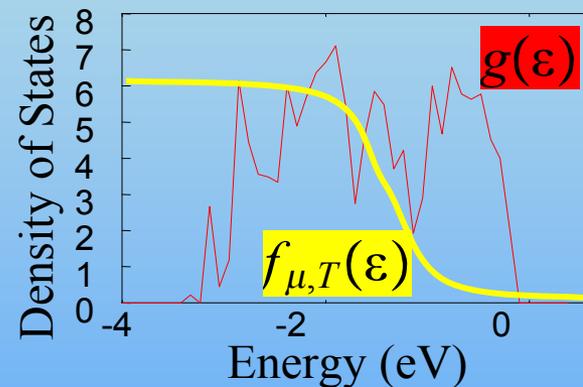
T-independent DOS and charge density

Electronic DOS

Fermi-Dirac Distribution

Electronic Free energy

Advantage: Get F_{elec} “for free”



$$F_{\text{elec}}(T) = E_{\text{elec}}(T) - E_{\text{elec}}(0) - TS_{\text{elec}}(T)$$

$$E_{\text{elec}}(T) = \int f_{\mu,T}(\epsilon) \epsilon g(\epsilon) d\epsilon$$

$$S_{\text{elec}}(T) = -k_B \int (f_{\mu,T}(\epsilon) \ln f_{\mu,T}(\epsilon) + (1 - f_{\mu,T}(\epsilon)) \ln(1 - f_{\mu,T}(\epsilon))) g(\epsilon) d\epsilon$$

Cluster expansion:

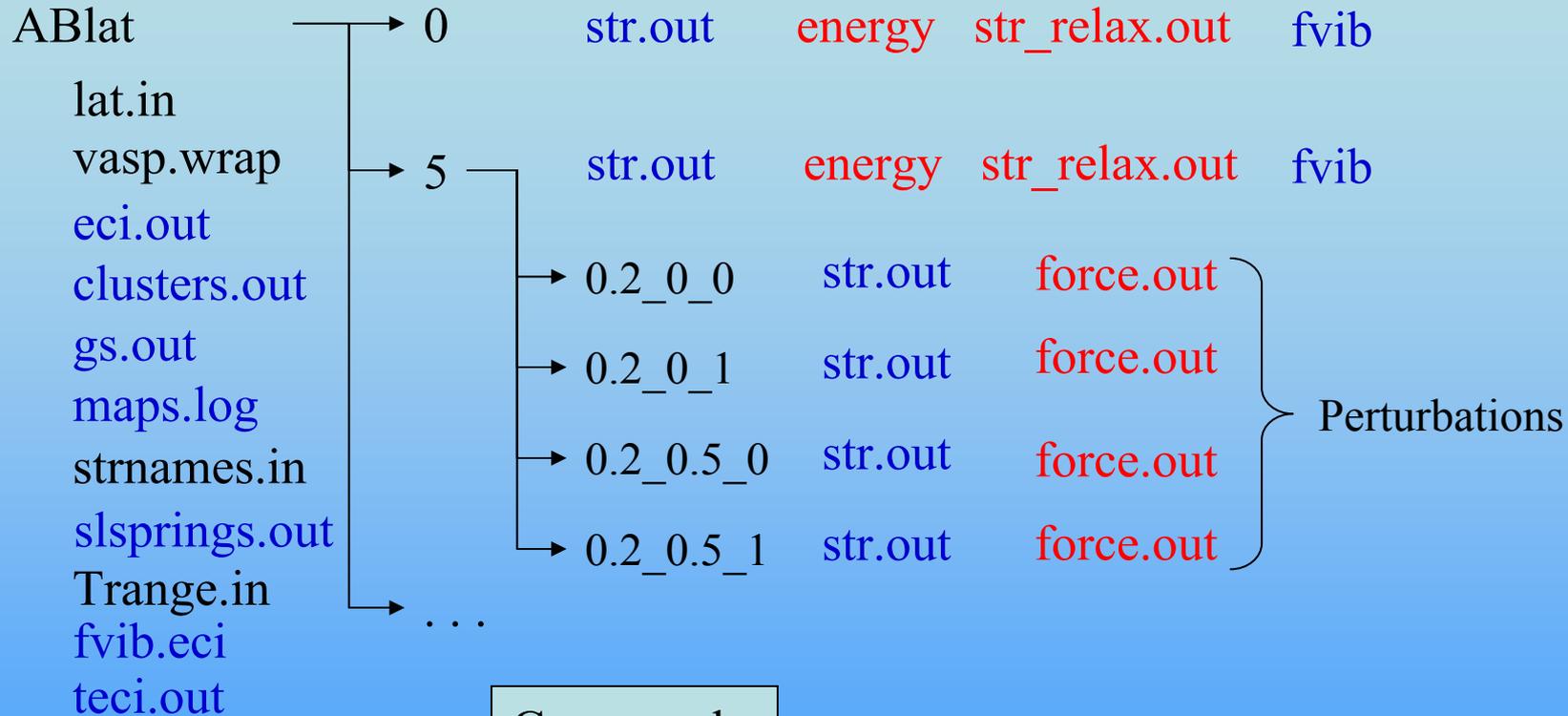
$$F(\sigma_1, \dots, \sigma_n) = \sum_{\alpha} J_{\alpha}(T) \sigma_{\alpha}$$

Using ATAT

- Overview of the input/output files
- Syntax of the files
- Sample output

File structure

Files:



Commands:

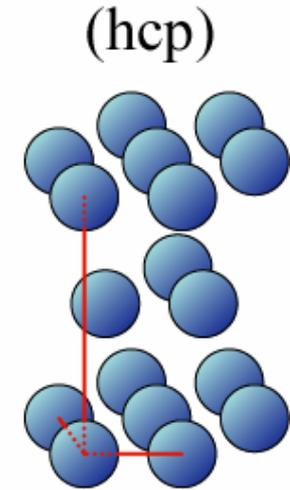
```
maps -d
run ab initio code
fitsvsl -d
run ab initio code

fitsvsl -f
foreachfile str.out svsl -d
clusterexpand fvib
mkteci fvib.eci
```

Example of input files

Simple lattice input file

	a	b	c	α	β	γ
Coordinate system	2.94	2.94	4.8	90	90	120
Unit cell	1 0 0	0 1 0	0 0 1			
Atoms	0.666667	0.333333	0.500000	Al, Ti		
	0.000000	0.000000	0.000000	Al, Ti		



Simple ab initio code input file

```
[INCAR]
PREC = high
ISMEAR = -1
SIGMA = 0.1
NSW=41
IBRION = 2
ISIF = 3

KPPRA = 1000
DOSTATIC
```

} Standard VASP tokens

{ k -point density
{ (k point per reciprocal atom)

do static run

Clusters and ECI files

clusters.out

eci.out

fvib.eci

teci.out

1
0.000000
0

2
0.000000
1
1.000000 1.000000 1.000000

multiplicity →

6
2.939999

diameter →

of points →

2
0.666667 0.333333 0.500000
1.000000 0.000000 1.000000

sites →

(using coordinate system in lat.in)

6
2.940004
2
0.666667 0.333333 0.500000
-0.333333 -0.666667 0.500000

etc.

0.229683
-0.350275
0.047938
0.041720

etc.

0.0826
-0.0008
-0.0006
-0.0005

T=0K

0.0793
-0.0008
-0.0006
-0.0006

T=100K

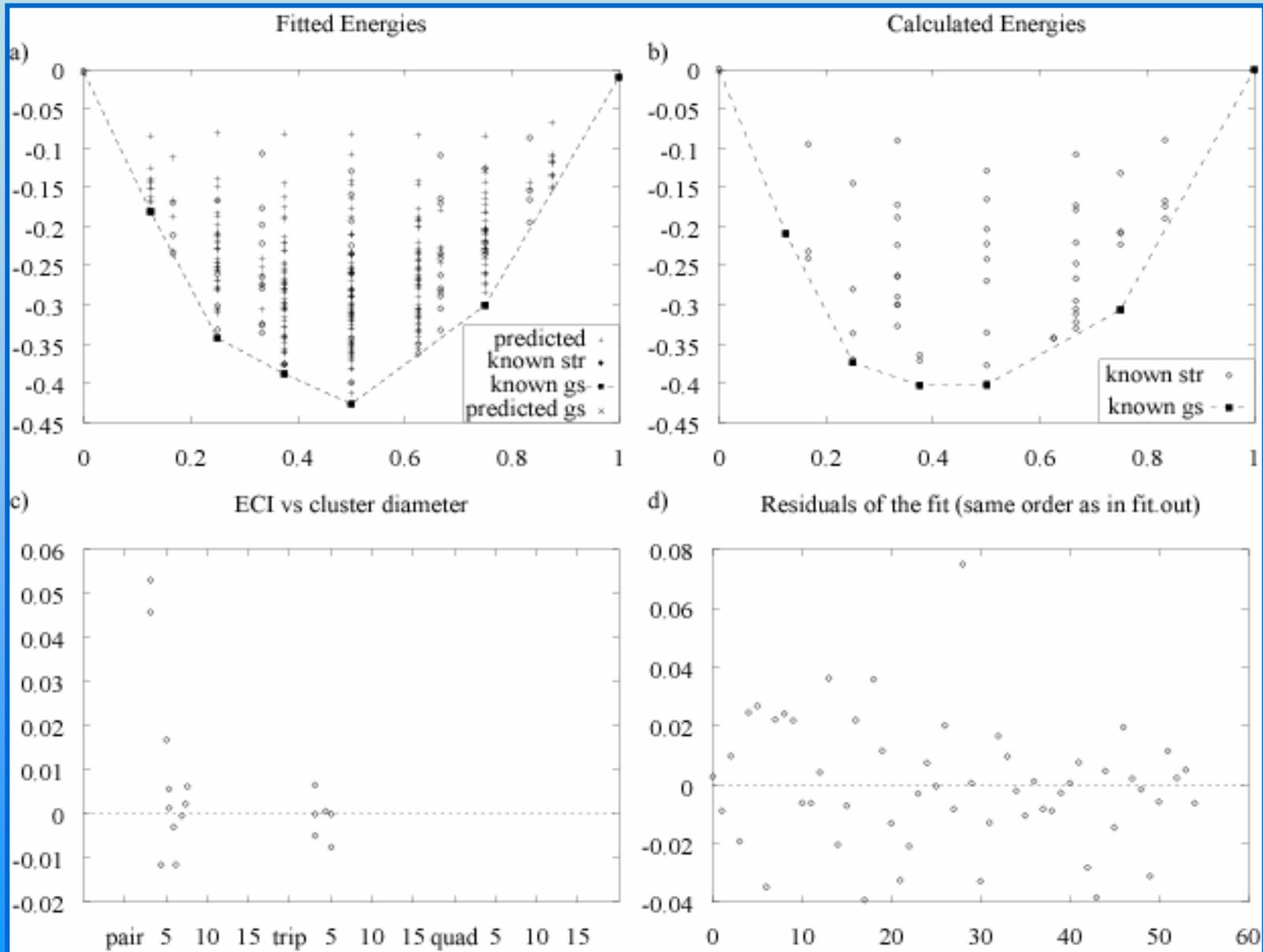
etc.

0.312346
-0.351077
0.0473349
0.0411423

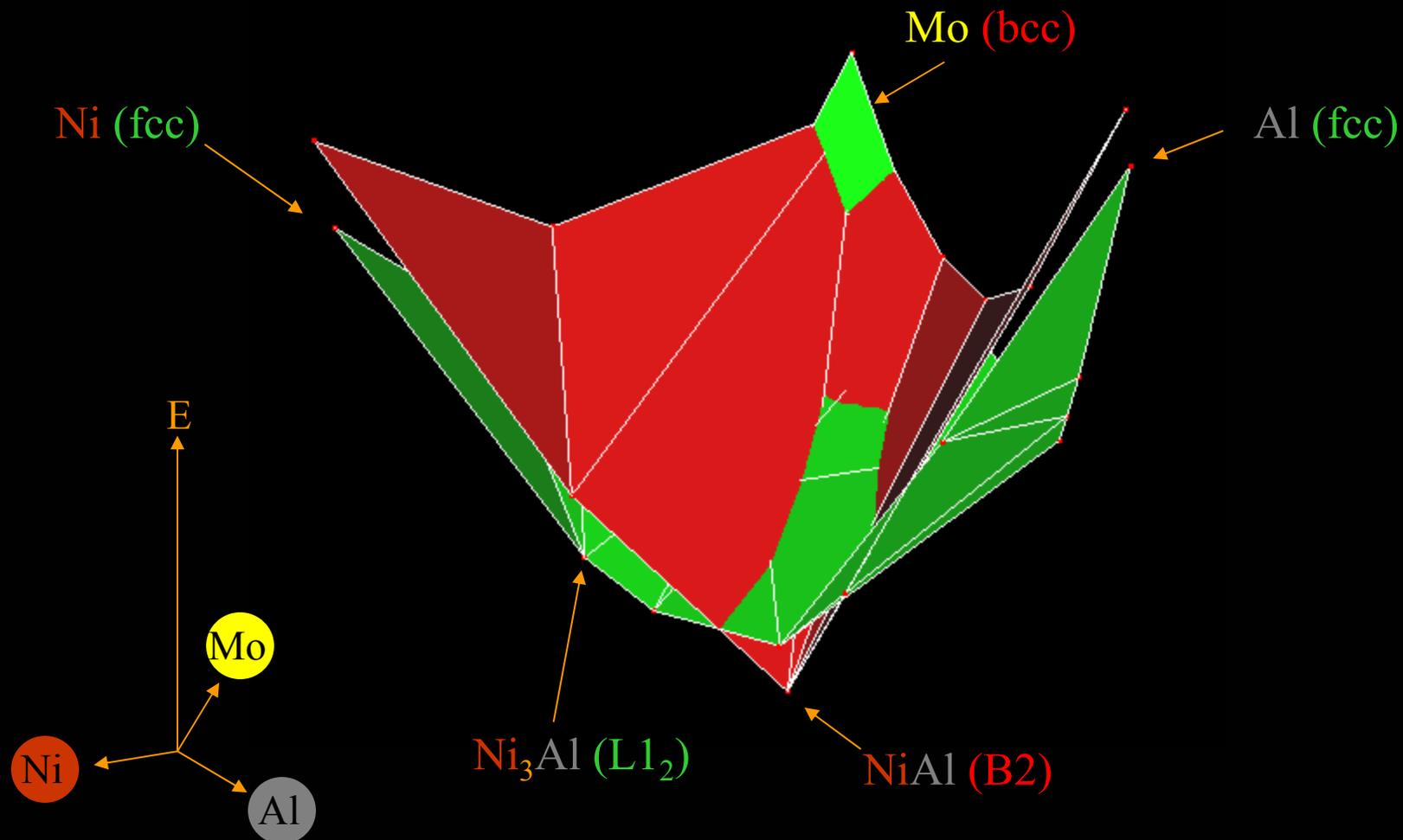
0.308527
-0.350903
0.0472538
0.0410874

T_{max} → 2000 21
division

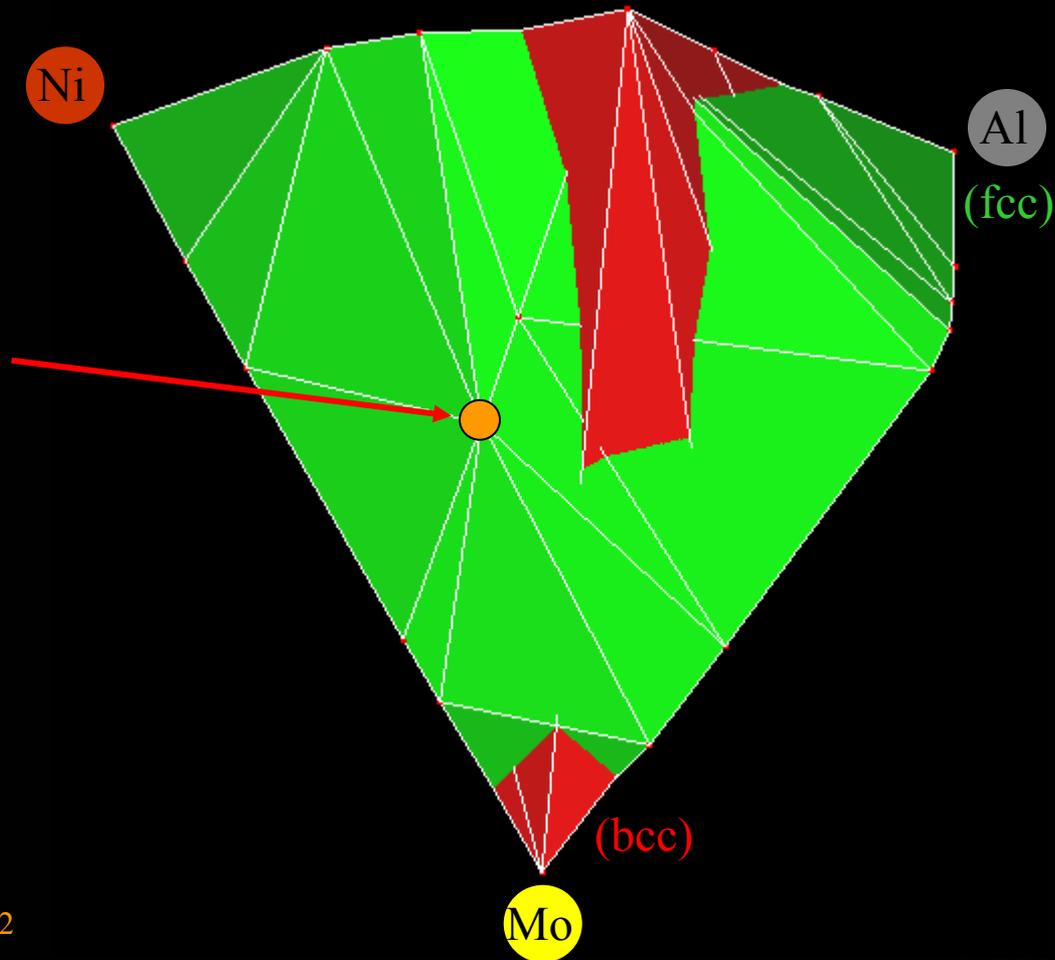
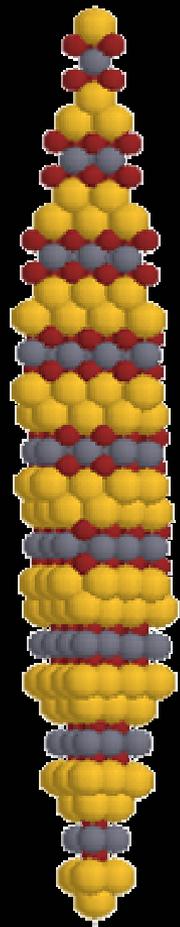
maps graphical output



Search for new compounds in the Al-Mo-Ni system



Predicted Compound

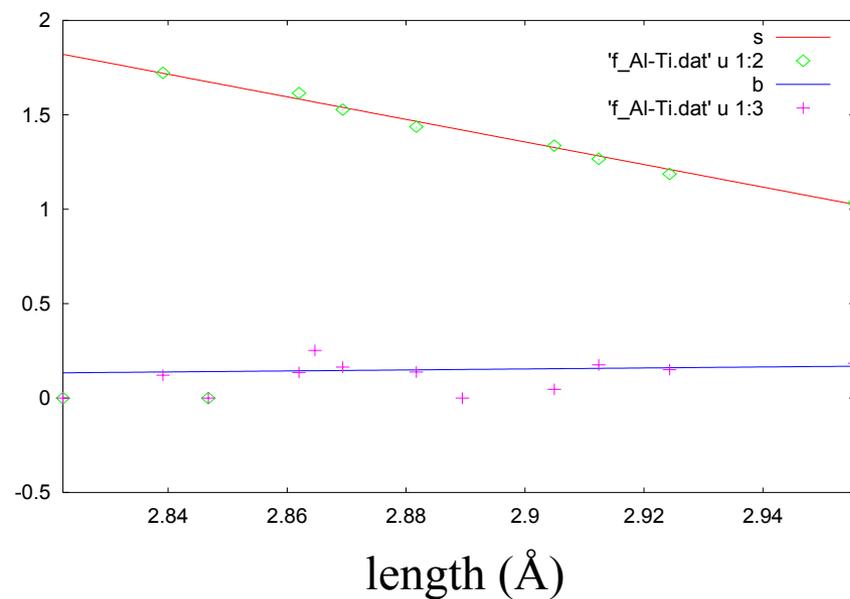
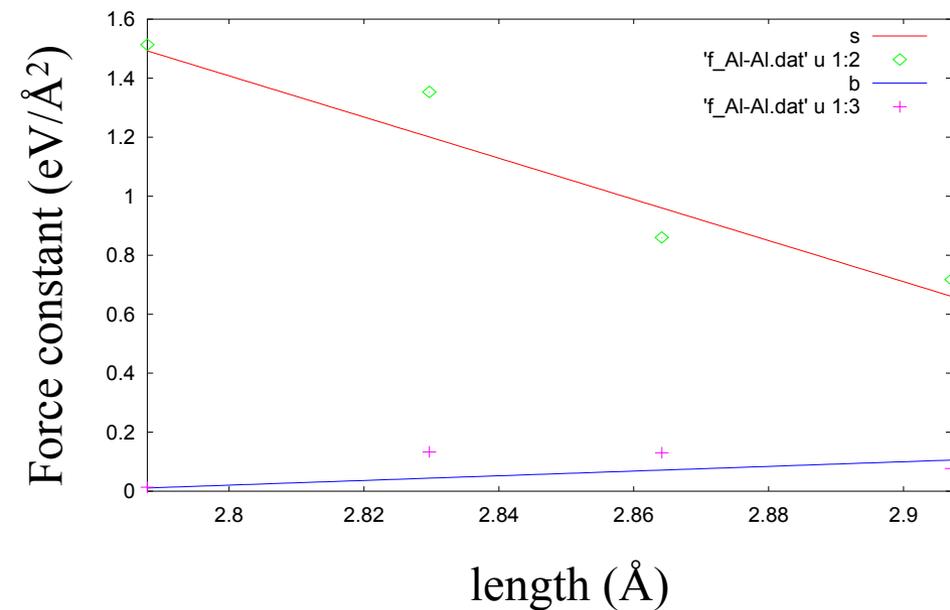


plotted with *MEDIT*, INRIA-Rocquencourt.

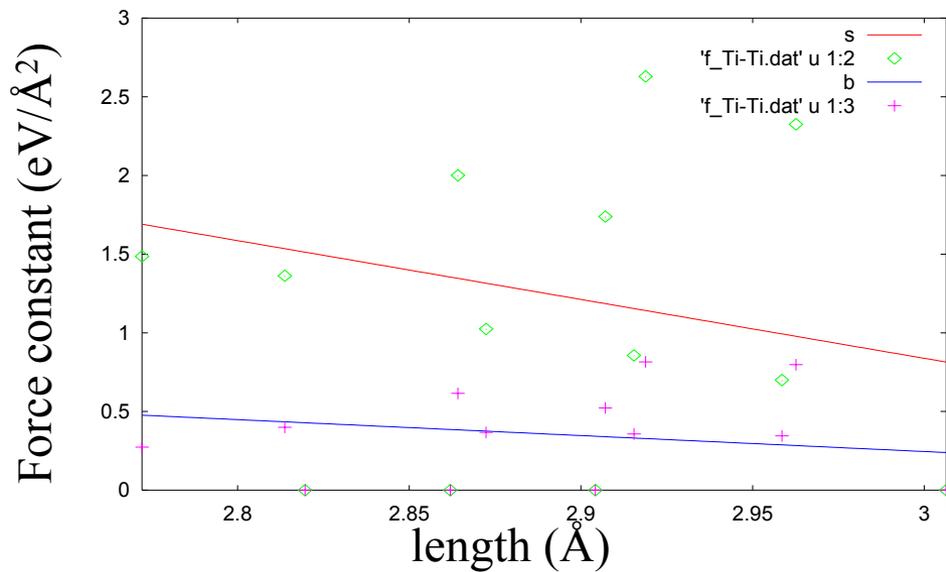
fitsvsl graphical output

Al-Al

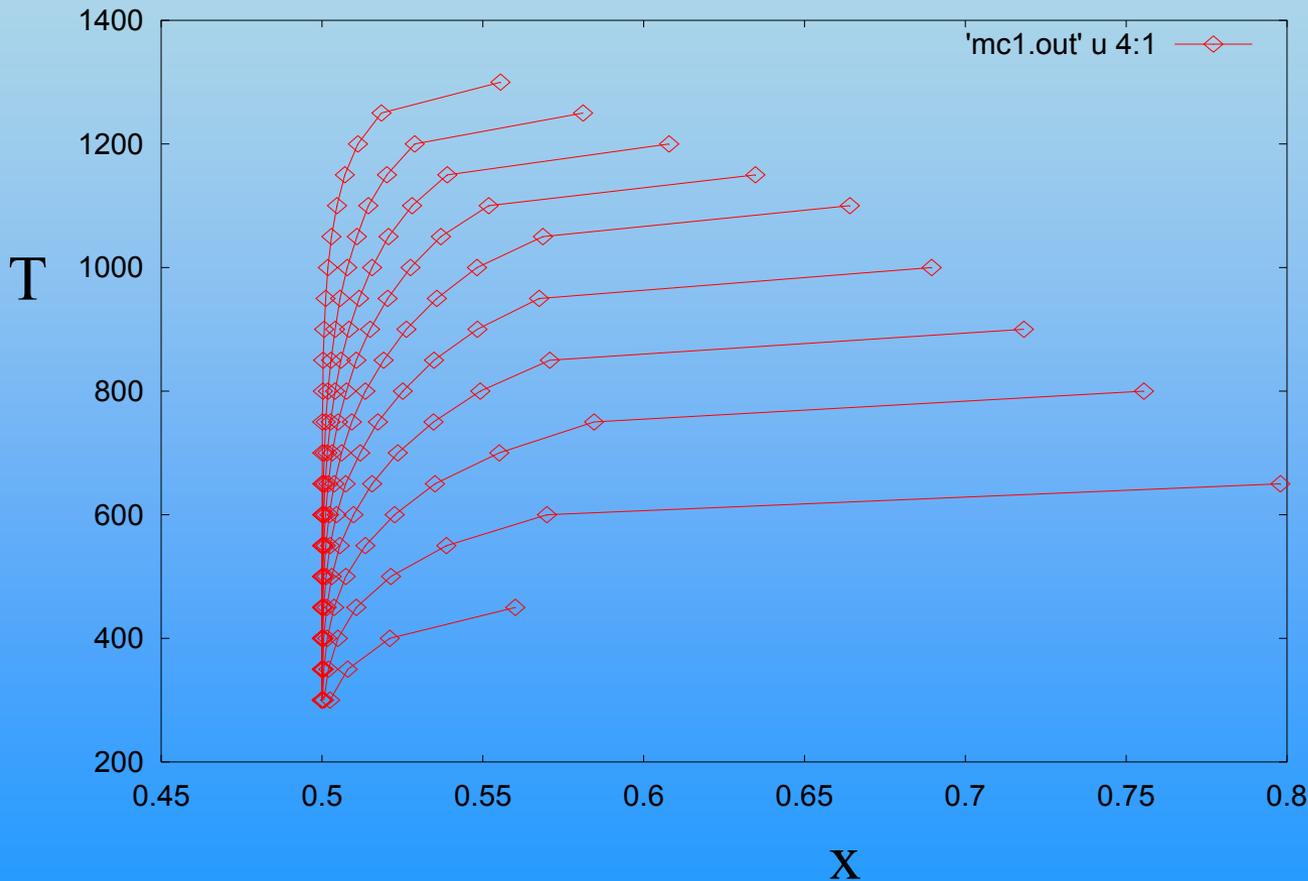
Al-Ti



Ti-Ti



emc2 graphical output



Other outputs:

- 1) T: Temperature
- 2) μ : chemical potential
- 3) $E - \mu * x$: Average energy
- 4) x: Average Concentration
- 5) ϕ : grand canonical potential
- 6) E2: heat capacity
- 7) x2: susceptibility
- 8) $E_{lte} - \mu * x_{lte}$: from LTE
- 9) x_{lte}
- 10) ϕ_{lte}
- 11) $E_{mf} - \mu * x_{mf}$: from MF
- 12) x_{mf}
- 13) ϕ_{mf}
- 14) $E_{hte} - \mu * x_{hte}$: from HTE
- 15) x_{hte}
- 16) ϕ_{hte}
- 17) lro: Long Range Order parameter
- 18-) corr: correlations

Miscellanea (I)

ATAT
Utilites

maps : Cluster expansion builder
mmaps : Multicomponent version of maps
emc2 : General purpose Monte Carlo code
phb : Phase-transition-tracing Monte Carlo code
checkrelax : Excessive relaxation detector
corrump : Cluster generator/correlation calculator
clusterexpand : Manual cluster expansion generator
genstr : Super structure generator
gensqs : Special Quasirandom Structure generator
pdef : Point defect supercells generator
csfit : Constituent strain calculator
cellcvrt : General crystal structure file format conversion utility
lsfit : Least-squares fitting code
fitfc : Phonon calculation with direct force method
fitsysl : Length-Dependent Transferable Force Constants generator
svsl : Phonon calculations using LDTFC
felec : Electronic free energy calculator

Miscellanea (II)

ATAT Utilites

`runstruct_vasp` : Interface with vasp
`runstruct_abinit` : Interface with abinit
`runstruct_pwscf` : Interface with pwscf } *ab initio*
`runstruct_gulp` : Interface with gulp ← empirical potential

`pollmach` : A job dispatcher for computer clusters
`foreachfile` : A “loop over directories” utility
`str2xyz` : File conversion utility for viewing with rasmol

`makelat` : Database of crystal structures

`getvalue`, `(just)after`, `(just)before`, `(just)between` `sspp` : Text extractors

`memc2` : Multicomponent version of emc2 (in development)

Getting Help

- Run command without any argument to get usage help
- Use `-h` option to get detailed help
- <http://cms.northwestern.edu/atat/manual/manual.html>

References

- A. van de Walle and G. Ceder. The effect of lattice vibrations on substitutional alloy thermodynamics. *Rev. Mod. Phys.*, 74:11, 2002.
- A. van de Walle, M. Asta, and G. Ceder. The alloy theoretic automated toolkit: A user guide. *CALPHAD Journal*, 26:539, 2002.
- A. van de Walle and G. Ceder. Automating first-principles phase diagram calculations. *Journal of Phase Equilibria*, 23:348, 2002.
- A. van de Walle and M. Asta. Self-driven lattice-model monte carlo simulations of alloy thermodynamic properties and phase diagrams. *Modelling Simul. Mater. Sci. Eng.*, 10:521, 2002.
- Web site: <http://cms.northwestern.edu/atat>
- This afternoon's tutorial: <http://cms.northwestern.edu/atat/tutorial>